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Fabrication of Prepackaged Superalloy Honeycomb Thermal Protection System (TPS) Panels

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FOREWORD

This is an interim report on work being performed by Rohr Industries, Inc., - Design and Fabrication of Titanium Multiwall Thermal Protection System (TPS) - describing the Task V activities. Task V, Concept Development of prepackaged Superalloy Honeycomb Sandwich panels consisted of:

- a. A material survey and preliminary design;
- Fabrication of component and full sized panels for structural and thermal tests;
- c. Thermal analysis;
- d. Structural analysis;
- e. Thermal and structural tests to verify the design analysis; and
- f. Fabrication of 25 panels for delivery to NASA Langley Research Center for additional testing.

This program is administrated by the National Aeronautics and Space Administration Langley Research Center (NASA LaRC). Mr. John Shideler of the Thermal Structures Branch, Loads and Aeroelasticity Division, is the technical monitor.

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SUMMARY

A material survey was conducted to find suitable materials that could be used as a Thermal Protection System (TPS) for one hundred missions on entry vehicles where the temperature range is 810° to 1,366°K (1,000° to 2,000°F) and pressure loads do not exceed 13.8 kiloPascals (kPa) (2 PSI). A combination of INCONEL 617, TI-6A1-4V and silica fiber materials were selected to be used as a sandwich. A TPS panel was designed using the thermal requirements for Space Shuttle Body Point 1300 as representative design criteria. Thermal and structural analyses were performed. Component specimens and full size panels were fabricated and tested to verify the design. Comparison of analytical and test data substantiate the analysis methods and verify the thermal and structural performance of the panels.

After design verification tests, one array of twenty panels, an array of two panels, and three single panels were fabricated and delivered to NASA Langley Research Center for additional testing.

1/ INTRODUCTION

As part of a program to develop lightweight durable Thermal Protection Systems (TPS) for future space transportation systems, titanium TPS panels have been studied for application where surface temperatures do not exceed 1000°F (References 1 through 5). This report describes an extension of the program to develop TPS for the temperature range from 1000°F to 2000°F. The objective of the work reported herein (Task V, Contract NASI-15646) was to survey high temperature materials and select a TPS material/configuration based on prepackaged superalloy concepts identified in References 5, 6 and 7, to analyze the selected design both thermally and structurally, and to fabricate and test specimens to obtain data for correlation with analysis. Finally, upon verification of the design, full-sized panels and arrays of panels were fabricated for delivery to NASA for additional testing.

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2/ DESCRIPTION OF CONFIGURATION

The design configuration is the result of various trade off studies performed on the original design supplied by NASA-Langley (see References 5-7). The trade offs involved the structural and thermal performance of the panel. The resultant design is shown in Figure 1-A.

The inner and outer layers of the panel are honeycomb sandwich. The original configuration called for dimpled core as used in the titanium multiwall concept. However, the use of honeycomb core was shown to be more efficient structurally and to be equivalent in thermal performance even though the honeycomb has a higher thermal conductivity. This is because the honeycomb core sandwich does not structurally require as much thickness as the dimpled core sandwich, and consequently there can be more fibrous insulation for a given panel thickness.

The side walls of the original design were slanted at 0.524 Radians (30 degrees) in an attempt to optimize thermal performance. Detailed investigation into this concept produced several objectionable features. First, since the center of pressure of the top layer of the panel did not line up with the centroid of the attachment clips, there was significant nonuniformity in the internal loading. Secondly, the sloped side walls were heavier and did not have the strength or stability of vertical sidewalls. Thirdly, finite element model studies revealed a thermal

kinematics problem between adjacent panels. With the sloped arrangement, adjacent panel sidewalls thermally grow and rotate into each other. Finally, a detailed thermal analysis showed the vertical sidewalls to have adequate thermal performance. As a result, the design configuration has vertical sidewalls which have corrugated flutes to provide stability and impede the flow of gases through the gap between panels during service.

The detail design Figure 1-A, employs a titanium 6A1-4V 4.32 mm (0.170 inch) thick honeycomb inner panel, a 7.11 mm (0.280 inch) thick Inconel 617 honeycomb outer panel with 12.7 mm (0.50 inch) thick Dynaflex and 35.31 mm (1.39 inches) thick Q Fiber Felt sandwiched between the two panels. The Inconel 617 honeycomb panel which was brazed includes two 0.13 mm (0.005 inch) thick skins, honeycomb core, and four side closures. The titanium 6A1-4V honeycomb panel which was Liquid Interface Diffusion (LID) bonded includes two 0.15 mm (0.006 inch) thick skins and honeycomb core.

The honeycomb core for the Inconel sandwich is 1/4 inch cell fabricated from 0.05 mm (0.002 inch thick) Inconel 617 foil. This foil thickness is the thinnest that can be brazed with the very aggressive braze alloy that was used. The cell size and face sheet thicknesses were determined by trade off studies which calculated the minimum weight of the sandwich system for the required strength. The critical strength parameter is intracell buckling. The core height is the minimum required to react the bending moment created by the pressure loads.

The honeycomb core for the titanium sandwich is 3/16 inch cell fabricated from 0.05 mm (0.002 inch) Ti-3A1-2.5V foil. This is the thinnest foil that can practically be LID bonded. The cell size, core height, and face

LID bonding is a Rohr Proprietary process in which the part interfaces are plated with one or more element which when heated to the proper temperature will melt, creating a short time eutectic melt with the titanium causing a bond to occur across the interface.

sheet thicknesses were determined by the same methods as those for the Inconel sandwich. The results of all stress analyses are discussed in Section 7.3 "Structural Analysis of Full Sized Panel."

The Q-Fiber Felt and Dynaflex were sized based on the predicted temperature range between the honeycomb panels. The Inconel 617 honeycomb panel was sized based on an entry temperature of $1,366^{\circ}$ K ($2,000^{\circ}$ F) and 2 psi external pressure load. The Titanium 6A1-4V honeycomb panel was sized based on the same entry conditions plus the concentrated load near each of the four corner attachment points.

The panels are normally attached to a vehicle by means of a bayonet and clip arrangement (see Figure 2). As shown, the bayonet goes into a clip on an adjacent panel and also through a vehicle clip. Thus each bayonet secures the forward edge of its panel, and the aft edge of the panel in front of it. The panel bayonets and clips are attached to the panels by diffusion bonding and the vehicle clips are mechanically fastened to the vehicle. The panels are installed in shingle fashion. Therefore, if a panel were damaged near the front of the vehicle, it would be necessary to start panel removal from the rear of the vehicle and remove an entire row to reach the damaged panel.

To have more flexibility in removal and replacement of panels on a vehicle, an alternate attachment concept was designed. It is a through-panel fastener concept and is used on a panel at the end of a row of panels. The through-panel fastener allows this end panel to be removed and access to be gained to the adjacent panels. As shown in Figures 1-A and 3, a sleeve structure with a removable cap is internally brazed to the panel and a bolt connects it to the vehicle substructure. The through-panel fastener was designed to transfer loads between the upper and lower panels and at the same time limit the through-panel thermal conductance. The conductance path is limited by the use of a plastic washer under the bolt head and by the small contact area between the bolt and lower panel. In addition, the fastener cavity is filled with fibrous insulation to limit direct radiation.

3/ DESIGN CONDITIONS

A Space Shuttle environment for body point 1300 was used as typical design criteria for this panel. The design point is located on the bottom centerline just aft of the cockpit. The design criteria for this panel included temperature and aerodynamic pressure environments for an ascent and a descent condition. These pressure loads and thermal gradients are tabulated in Table 1.

The ascent condition provided the maximum pressure load (ΔP) on the panel. This load was contractually set at 14 KPa (2.0 psi) ultimate. Accurate determination of a typical pressure load for the panel is difficult because such loads can be associated with complex surface pressure gradients which occur due to shock waves on the vehicle surface. However the 14.0 KPa (2.0 psi) agrees well with that derived during the Reference 7 study. This study included two areas which are also on the underbody of the shuttle and have temperature environments similar to BP 1300. One is designated Area II and is located on the lower aft fuselage. The other is designated Area III and is located on the main landing gear door. The associated surface temperature gradients for the 14.0 KPa (2 psi) design load was conservatively assumed to be the maximum one of either Area II or Area III. This turned out to be Area II and is shown on Page 2-9 of Reference 7 and Table 1 of this report.

The descent condition provided the maximum thermal environment and thermal gradient. The temperature and pressure data tabulated in Tables 2 and 3 were used to calculate the temperature distributions shown in Figure 4. The critical thermal gradient occurred at time = 500 seconds where the outer surface reaches its maximum temperature value of 1900°F. At this time, the inner surface is still relatively cool at 208°F so the maximum temperature gradient exists on the panel. Reference 7 study showed that there are not any pressure loads on the Area II and Area III panels during these elevated temperature exposures. The shock pressures are exerted after the panels have cooled down to near ambient temperature. These two conditions, providing separately the maximum pressure and thermal gradients on the panel, are used in the Section 7.3 structural analysis.

The effects of time at temperature were also considered during the test program and during the stress analysis. Basically, this consideration is that the panels during entry are exposed to 1256° to 1366°K (1800° to 2,000°F) environment for approximately 300 seconds during every flight or approximately 8 hours for 100 flights.

4/ MATERIAL SURVEY

Literature was searched to locate a suitable metal that would retain adequate strength at temperatures up to 1,366°K (2,000°F) for 100 hours. This selection was based on the fact that a 100-mission reuse requirement for TPS for a shuttle type vehicle represents a total life requirement on the order of 10 to 100 hours at elevated temperature.

At elevated temperatures, the short time mechanical properties (F_{tu}, F_{ty}) are still of importance in design, but time-dependent properties become the governing design consideration. Creep strength, metallurgical stability, and oxidation resistance are included in this category. The creep strength of an alloy will determine its high temperature load-carrying ability while oxidation will have to be accounted for by an increase in thickness to maintain the required load carrying capability for the total life. In addition to the above criteria, availability, cost and fabricability have to be taken into account in determining the most suitable alloy. Material in the following gauges were required for this task:

- a. 0.051 mm by 102 mm wide (0.002 inch) by 4 inches wide)
- b. 0.076 mm by 330 mm wide (0.003 inch by 13 inches wide)
- c. 0.127 mm by 330 mm wide (0.005 inch) by 13 inches wide)

Four alloy families were considered. They are:

- a. Precipitation strengthened (PH) super alloys
- b. Oxide dispersed alloys
- c. Refractory alloys
- d. Solid solution strengthened alloys

4.1 PRECIPITATION STRENGTHENED (PH) SUPERALLOYS

Gamma prime, the main strengthening precipitate of Precipation Strengthened Superalloys, starts to become metallurgically unstable after short exposures to temperatures at or around 1,366°K (2,000°F). This instability (overaging or solutioning) is reflected in the degradation of high temperature mechanical properties. This family of alloys must therefore be excluded from consideration. Rene 41 (see Table 4), for example, has been considered in previous studies (Reference 5) as a potential TPS material. The solutioning temperature of Rene 41, however, is 1,338°K (1,950°F). Exposure of this material to 1,366°K (2,000°F) would thus result in a material with extremely low creep strength that would be totally unsuitable.

4.2 OXIDE DISPERSED (OD) ALLOYS

The OD alloys such as thoria dispered (TD) nickel, TD nickel-chromium, and MA 956 [Yttria (Y_2O_3) dispersed] have adequate 1,366°K (2,000°F) yield and creep strengths (see Table 4). However, use of these alloys may result in fabrication and availability problems. MA 956, for example, has only been rolled to 0.012 inch. The TD nickel alloys have over 12-month lead times and cannot be rolled down to the required dimensions indicated in Reference 3 at this time.

4.3 REFRACTORY ALLOYS

Refractory materials such as columbium, molybdenum, and tungsten alloys have more than adequate $1,366\,^{\circ}\text{K}$ ($2,000\,^{\circ}\text{F}$) yield and creep strengths. However, they are inherently difficult to use in fabrication processes, require a coating to protect them from oxidation at high temperature, and

they become brittle at room temperature. Due to the encountered difficulties, this family of alloys is usually considered as TPS material for temperatures above 1,366°K (2,000°F) only.

4.4 SOLID SOLUTION STRENGTHENED SUPERALLOYS

Solid solution strengthened alloys, as the name implies, receive much of their high temperature strength from solute refractory (chromium, molybdenum, tungsten) and cobalt atoms. These atoms strengthen by acting to retard dislocation movement. In addition, these alloys are also strengthened through carbide precipitation.

In selecting a suitable candidate TPS material, one of the most useful sets of data for comparison purposes is the $1,366^{\circ}K$ ($2,000^{\circ}F$) 100 hour 0.2 percent specific creep strength, which may be derived from the 100 hour 0.2 percent creep strength. Unfortunately, this data is not as readily available for all the potential solid solution strengthened superalloys as is the $1,366^{\circ}K$ ($2,000^{\circ}F$) 100 hour creep rupture data. The main set of data used in comparing the creep behavior of the differing alloys was therefore the creep rupture data.

A list of candidate solid solution strengthened superalloys is shown in Table 4. As creep strength to weight ratios are important for any high temperature aerospace component, the alloys in Table 4 are listed 1 to 10 in order of their $1,366^{\circ}K$ ($2,000^{\circ}F$) 100 hour creep rupture specific strength. $1,366^{\circ}K$ ($2,000^{\circ}F$) and $1,255^{\circ}K$ ($1,800^{\circ}F$) creep rupture (100 hour) and short time Ultimate Tensile Strength (UTS) results are also shown for comparison.

As can be seen in Table 4, the three alloys that stand out as having exceptional 1,366 $^{\circ}$ K (2,000 $^{\circ}$ F)/100 hour creep rupture specific strength are INCOLOY® 802, INCONEL® 617 and L605. The 1,366 $^{\circ}$ K (2,000 $^{\circ}$ F)/100 hour

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creep rupture strengths of approximately 20 MegaPascals (MPa) (3.0 kilopounds per square inch (ksi)) of these alloys are from 30 percent to over 200 percent greater than the creep rupture strengths of the rest of the solid solution strengthened alloys listed in Table 4. Approximately the same ratios also hold true for the creep rupture specific strengths.

Although INCOLOY 802 can be rolled down to sheet, it is not available commercially in the thin gauges required. Likewise, L605 is unsuitable because:

- a. It has poor oxidation resistance $[1,255^{\circ}K (1,800^{\circ}F)/100 \text{ hour oxidation loss of } 0.0889 \text{ mm } (0.0035 \text{ inch})]$ (Reference 8), and
- b. It contains 53 percent Cobalt which increases costs and lead times.

INCONEL 617 is available in the required gauges and has excellent oxidation resistance. INCONEL 617 was therefore selected as the candidate material.

4.5 INCONEL 617

INCONEL 617 is a solid-solution, Ni-Cr-Co-Mo alloy with an exceptional combination of high temperature strength [100 hour, 1,366°K (2,000°F) 0.2 percent creep strength of 10.3 MPa (1.5 ksi)] and resistance to 1,366°K (2,000°F) cyclic oxidation (References 7, 9, 10, 11, and 12). Due to its exceptional properties, it is currently used in the combustion section of gas turbines. Strengthening of the alloy during exposure to temperature originates primarily from discrete $\rm M_{23}$ $\rm C_6$ precipitates. This phase was found to remain stable at temperatures up to 1,366°K (2,000°F).

INCONEL 617 has good fabricability and formability. Machining and welding are carried out using standard procedures for nickel alloys.

^{*} HASTELLOY is a registered trademark of Satellite Division, Cabot Corportaion, Kokoma, Indiana.

5/ FABRICATION

- 5.1 FABRICATION OF HONEYCOMB SANDWICH COUPON TEST SPECIMENS Specimens of INCONEL 617 were fabricated for testing in:
 - a. Face sheet tension
 - b. Creep
 - c. Edgewise compression
 - d. Flatwise tension
 - e. Pressure/Thermal Gradients
 - f. Thermal conductivity.

All honeycomb sandwich panels were fabricated 7.1 mm by 304.8 mm by 304.8 mm (0.280 inch by 12 inches by 12 inches) and subdivided into the appropriate test specimen sizes. A modified brazing/diffusion bonding process was used for joining the INCONEL 617 honeycomb panels. The process consisted of applying braze alloy (1.97B-0.02C-13.13Cr-3.4Fe-Ni Balance) approximately 40 grams per square foot to one side of each face sheet, and installing 6.35 mm (0.250 inch) cell honeycomb core between the face sheets for joining.

The layup was placed on a flat reference in a vacuum furnace where 0.14 kilograms (0.3 pounds) per square inch of tungsten pellets were added on top of the panels to provide pressure for brazing and diffusion

bonding. The furnace was then evacuated to 1 \times 10⁻⁴ torr and heated to 1,450°K (2,150°F), held for three minutes, then cooled to 1,311°K (1,900°F) and held for one hour before cooling to 431°K (300°F) and removing from the furnace. After bonding, all honeycomb specimens were evaluated using the ultrasonic through-transmission C-scan method.

5.2 FABRICATION OF FULL SIZE PANELS FOR PRESSURE AND THERMAL CONDUCTIVITY TESTS

INCONEL 617 subassemblies and titanium subassemblies were fabricated separately and then joined in a third assembly process.

5.2.1 FABRICATING THE INCONEL 617 SUBASSEMBLY -- The 0.13 mm by 313.30 mm by 313.30 mm (0.005-inch by 12.334-inch by 12.334-inch) skins were square sheared. The honeycomb core 6.35 mm (0.25 inch) cell by 0.08 mm (0.002 inch) thick foil by 304.8 mm by 304.8 mm by 101.6 mm (12 inches by 12 inches by 4 inches) was fabricated using a Rohr Coremaster machine. The 101.6 mm (4-inch) log was subdivided into 304.8 mm by 304.8 mm by 7.11 mm (12-inch by 12-inch by 0.280-inch) pieces using an electric discharge saw and a conventional mill and belt sander.

The side closures were formed on the 195-255 form tool as shown in Figure 5 and then hand trimmed. Since INCONEL 617 is relatively easy to form at room temperature, the form tool was made of 6061 aluminum. This form tool was machined using the numerical control machining process and then hand sanded to a smooth finish. The parts were formed in an ASEA hydropress. The side closures were formed in two stages. In the first stage, the corrugations were formed in the 195-256-9, -11, -13, and -15 side closures. (See Figure 1B for part numbers.) In the final stage, one insert was removed from each end of the form tool and one insert was added to each side of the form tool for forming the end flanges on the -13 and -15 side closures. Figure 6 shows the finished form tool and tool proof parts.

All parts were process cleaned in a pickling solution of nitric/ hydrofluoric acid before assembly. The parts were assembled with braze alloy (1.97B-.020C-13.13CR-3.45Fe-Ni Balance) applied at all interfaces as shown in Figure 7. All components were resistance spot tack welded together at each joint. This made the assembly shown in Figure 8 self supporting for brazing/diffusion bonding. Brazing/diffusion bonding was accomplished in a vacuum furnace at a pressure of 1 X 10⁻⁴ torr and temperatures of 1,450°K (2,150°F) for three minutes, then cooled to 1,311°K (1,900°F) and held for one hour. After bonding, all honeycomb-core-to-skin joints were evaluated using the ultrasonic through-transmission C-scan method.

5.2.2 FABRICATING THE Ti-6Al-4V SUBASSEMBLY -- The Ti-6Al-4V skins were designed with flanges on two sides of each skin which close out the sides of the Ti-6Al-4V subassembly. Due to this configuration and the thin gage 0.15 mm (0.006 inch) material, a superplastic forming process was selected. The superplastic forming tool shown in Figure 9 was designed to form the outer and inner skins simultaneously. Forming was accomplished in a vacuum furnace where a protective environment could be provided while forming the thin gage titanium.

C1020 steel was selected as the tooling material based on the coefficient of thermal expansion and the small number, approximately 25 each, of parts required for this program. Figure 9 shows tool proof parts being removed from the tool.

The honeycomb core was fabricated in a log of 304.8 mm by 304.8 mm by 101.6 mm (12 inches by 12 inches by 4 inches) by 4.7 mm (0.18 inch) cell size by 0.05 mm (0.002 inch) foil gage, using the Rohr Coremaster machine. The core log was then subdivided into 4.3 mm (0.17 inch) thick pieces. The core was plated for LID bonding using a Rohr proprietary process.

Final cleaning was accomplished by immersion in a vapor degreaser. LID bonding was accomplished in a vacuum furnace that was evacuated to 1 \times 10⁻⁵ torr. The part was heated to 1,213°K (1,725°F) and held for a period of time while LID material was being diffused into base material

to make the joints. After bonding, all honeycomb core to skin joints were evaluated using the ultrasonic C-scan method. Figures 10 and 11 show the completed titanium subassembly.

5.2.3 JOINING THE SUBASSEMBLIES -- The flanges of both subassemblies were prepared for LID bonding of the bi-metal joint using a Rohr proprietary process. After preparation for LID bonding, the INCONEL 617 subassembly was filled with 12.7 mm (0.5 inch) of precut DYNAFLEX and 35.3 mm (1.39 inch) of precut Q-FIBER FELT, as shown in Figures 12 and 13. After the DYNAFLEX and Q-FIBER FELT had been installed, the titanium subassembly shown in Figure 10 was installed over the Q-FIBER FELT. The flanged areas of both subassemblies were then resistance spot tack welded to each other for LID bonding. Since the subassemblies were resistance spot tack welded to each other, the assembly was somewhat self-fixturing. Only a flat reference surface was required to support the panel for LID bonding. Figure 14 shows the assembly being laid up for LID bonding the bi-metal joint. Figures 15 and 16 show a completed bi-metal panel with bayonet/clip attachments.

The 59.7 mm by 304.8 mm by 304.8 mm (2.35-inch by 12-inch by 12-inch) panel with clips and tongues weighed 0.926 kilograms (2.04 pounds). The same size panel with only through-panel fasteners weighed 0.898 kilograms (1.98 pounds). All panels were checked dimensionally and visually for defects.

The 195-254 through-panel fastener (Figure 3) is fabricated as a braze/diffusion bonded assembly. The base, flange and housing are fabricated using a standard production type blank die. The threaded insert and cap are machined using a hand screw machine (turret lathe). The parts are cleaned for brazing using a degrease solution. These parts are then assembled and resistance spot tack welded into position. Braze alloy (1.97B-0.02C-13.13Cr-3.4Fe) is applied at each joint and the assembly is placed in a vacuum furnace with the flange side down for braze/diffusion bonding at 1,450°K (2,150°F). Only a visual inspection is required to determine quality.

- 5.2.4 FABRICATION OF PANEL ARRAYS -- A twenty-panel array, a two-panel array, and three separate panels were fabricated and delivered to NASA Langley Research Center for further testing.
- 5.2.4.1 Twenty-Panel Array -- The twenty-panel array was designed to fit an existing 1078.5 mm by 1523.0 mm (42.46-inch by 59.96-inch) opening in the test apparatus for the 8-foot High Temperature Structures Tunnel. The basic panel size is 304.8 mm by 304.8 mm (12.0 inches by 12.0 inches). Therefore, three panels of 284.2 mm by 304.8 mm (11.19 inches by 12.0 inches), one panel of 284.2 mm by 149.4 mm (11.19 inches by 5.88 inches) and four panels of 304.8 mm by 149.4 mm (12.0 inches by 5.88 inches) in addition to twelve basic panels were required to fill the test fixture. An individual panel is shown in Figure 16 and the twenty-panel array is shown in Figure 17.

The panel joints were aligned with the flow so that gas flow in the joints could be studied during tunnel tests. The array of panels were attached to a 4.8 mm (0.190 inch) thick plate, shown in Figure 18, which represents the mass of the shuttle fuselage structure at the design location, body point 1300.

Panel fabrication was accomplished using the process parameters described in Section 6. The panels were processed six at a time, as shown in Figure 19. The quantity was governed only by the available furnace size.

All honeycomb subassemblies were evaluated using the ultrasonic through-transmission C-scan method. All subassemblies and final assemblies were checked dimensionally for conformance to the drawing. The final assemblies, such as that shown in Figure 15, were pressure checked in an unrestrained position to 14 KPa (2 psi) internal pressure.

To pressure check the panels a Meriam manometer using Meriam 295 Red Fluid (2.95 specific gravity), shown in Figure 20, was used. A regulator

in the airline was used to prevent the panel from being over-pressurized when the flexible tygon line was placed over the vent hole in the lower panel.

Evaluation showed some panels to have intracell dimpling of the face sheets. This was not considered to be a structural problem since some of the specimens tested and reported in Section 7 had intracell dimpling and had acceptable test results.

- 5.2.5 INSTRUMENTATION -- The 20-panel array and the 2-panel array were instrumented with Type K thermocouples. INCONEL sheath was used where the temperature was expected to be above 1,255°K (1,800°F) and 30 gage fiberglass sheath couples were used in areas where the temperature was expected to be below 1,255°K (1,800°F). Five INCONEL sheath type couples were installed inside an INCONEL 617 subassembly before final assembly. This panel was installed at the 2-C location in the 20-panel array. Figures 21 and 22 show the thermocouple layout for both arrays.
- 5.2.6 INSTALLATION -- The panels having clips and tongues as means of attachment were somewhat more difficult to install on the aluminum plate than the panels having through-panel fasteners. This was due to having to compress the NOMEX felt, which was coated with RTV rubber, while sliding the tongue into the clips. The 20-panel array had pressure probe connections installed in seven places, as shown in Figure 21. The pressure probes were located to detect pressure buildup between the aluminum plate and the bottom side of the panels during tunnel tests.

Three additional panels were mounted on individual 4.8 mm (0.19 inch) thick plates. These plates each had NOMEX felt installed between the panel and the plate, but had no instrumentation. These panels were interchangeable with other panels in the 20-panel array.

6/ THERMAL PERFORMANCE

The procedure followed for the thermal analysis was:

- a. Entry conditions were used for shuttle body point 1300 and a transient thermal analysis was run to size insulation thickness, (i.e., design of overall tile thickness).
- b. Steady-state temperatures were measured across manufactured tiles. Measured hot and cold face surface temperatures were used and a steady-state thermal analysis was performed to predict temperatures and effective conductivity, and to correlate them with test values.
- Figure 23 represents the thermal math model used in the MITAS lumped parameter thermal analysis computer program (Reference 13) to size the insulation thickness of the tile. The temperature and pressure histories shown in Tables 2 and 3 for shuttle body point 1300, trajectory 14414.1C were supplied by Langley Research Center as a starting point for the thermal analysis. Thermophysical properties of the INCONEL 617 honeycomb, DYNAFLEX®, Q-FIBER FELT®, titanium honeycomb, and aluminum used in the analysis are provided in Tables 5-8, respectively.

DYNAFLEX is a registered trademark of Johns-Manville Corp., Denver, Colorado.

Q-FIBER FELT is a registered trademark of Johns-Manville Corp., Denver, Colorado.

Because the external pressure varies with time, the insulation thermal conductivity was allowed to vary with pressure in addition to the usual temperature variation. The resulting maximum temperature of the aluminum plate was determined from a transient analysis for various cases, with a different, arbitrarily selected, thickness of insulation. To ensure that stored energy at the end of entry does not continue to heat the aluminum structure, the given temperature and pressure histories were extended to 2,000 seconds. Because the Q-FIBER FELT has a temperature limit of 1,255°K (1,800°F), care was taken in the analysis to ensure that this limit would not be exceeded.

During the early computer runs, the temperature at point 28 of the math model was monitored and evaluated as a function of DYNAFLEX and Q-FIBER FELT thicknesses. From this it was determined that a 12.7 mm (0.5 inch) thickness of DYNAFLEX would keep the insulation interface below 1,255 K (1,800°F). The remaining analyses, therefore, had the DYNAFLEX thickness fixed at 0.5 inch, but used various Q-FIBER FELT thicknesses.

Typical results of the transient analysis are shown in Figure 23, where temperature responses are shown for the case where the total thickness of the tile is 55.9 mm (2.35 inches). For this case the insulation interface was 1,117°K (1,550°F) and the maximum temperatures of the aluminum was 439°K (330°F). The maximum temperature of the aluminum for three insulation thicknesses is shown in Figure 24. Extrapolation of this data to 454°K (350°F) establishes a required thickness of 58.4 mm (2.3 inches). Because the TPS thickness was selected at an early stage in the program, the tile design thickness of 59.7 mm (2.35 inches) was not changed.

- 5.2 STEADY-STATE ANALYTICAL PREDICTION AND CORRELATION OF TEMPERATURE DISTRIBUTION AND EFFECTIVE CONDUCTIVITY WITH THE TEST RESULTS FOR THE BI-METAL THERMAL PROTECTION SYSTEM
- 6.2.1 TESTS -- Thermal conductivity tests were performed using a modified guarded hot plate shown in Figures 25 and 26. The hot plate has quartz lamps that are divided into three independent heating zones;

control, mid, and edge. Separate automatic controls are used to minimize the temperature gradient between the central test section and the mid guard heater. The edge guard heater, in turn, minimizes the temperature gradient between the mid test section and the edge. In this way, the apparatus is a double guarded system. This minimizes any radial heat flow away from the central test section. Min-K, having a known thermal conductivity, was used as a test standard to calibrate the test apparatus and run thermal conductivity tests.

The test panels are shown in Figures 27 and 28. The test setup shown in Figure 29 was used for checking thermal conductivity of the superalloy panel. The test panel was placed on top of a honeycomb panel and the known thermal conductivity instrumented Min-K was placed on top of the test panel. The honeycomb panel was used as the "Hot Plate" to provide a more uniform heating of the test specimen. The honeycomb panel was instrumented with thermocouples, the outputs of which were fed into the automatic control circuit in order to maintain the test temperature. The test panel was instrumented with thermocouples that were welded onto both sides of the panel surface at the center, midway between the center and edge, and at edge locations. Because of the physical nature of the Min-K. thermocouples could not be attached directly to its surface. Therefore, thermocouples were put on small INCONEL rectangular tabs which were insulated from the metal surfaces of the test panel and aluminum plate, but were forced onto the Min-K surfaces by the weight of the test setup. Thermocouple plan-form locations on the Min-K were the same as for the test panel.

6.2.2 ANALYSES -- Measured boundary temperatures obtained from the steady-state thermal conductivity tests were used as boundary conditions in thermal analyses to predict the temperature distribution of a 304.8 mm by 304.8 mm (12- inch by 12-inch) bi-metal TPS tile.

Figure 30 represents the thermal math model used for the analysis. The model differs from the transient model in that it includes a cold face boundary, Min K insulation, no primary structure, and no gap radiation.

The thermophysical properties used for the analysis are presented in Tables 5 through 9. Since the test was conducted at sea level pressures, only the 2,116 pounds per square foot (1 atmosphere) were used.

Results

Steady-state computer runs were performed using, as boundary temperatures, the measured temperatures of the hot face (node 1) and cold face (node 17). The analytical temperature of node 13 (cold side of the titanium honeycomb) was compared with the measured temperature. Two sets of computer runs were performed, one without sidewall to predict the temperature in the center of the tile, and another with a sidewall to predict the temperature adjacent to the sidewall. The transient model and the original steady state model had the sidewall conducting directly from node 1 to node 13. It was necessary to change the model only for one steady state solution. That is, measurements along a line directed from hot face to cold face and through the tile center could be correlated without sidewall conduction in the thermal model. Measurements near the sidewall needed the addition of sidewall conduction retained in the thermal model to obtain a close correlation. Table 10 presents the boundary temperatures used and a comparison of the predicted and measured temperatures for node 13.

Figure 31 presents the percent error of the predicted temperatures versus measured temperatures. For the area above the zero percent line, the analytical model predicts higher temperatures and is, therefore, conservative. Based on that error, a 452°K (350°F) analytical predicted temperature for the cold face of the titanium honeycomb will have an actual temperature of 447°K (344.8°F). Based on the sidewall error

curve, the cold face of the titaninum honeycomb will be 422°K (335.7°F). In the actual case, the aluminum structure diffuses the temperature so that actual temperature will be somewhere between the two.

Figure 32 presents the effective thermal conductivities (calculated from temperatures obtained from the test data and from temperatures obtained from the thermal math model) as a function of mean temperature at the center of the panel.

The center measured temperature differences (ΔT) and thickness (1) of the test specimen (TS) and Min-K (MK) were used to calculate the effective thermal conductivity (k) as follows:

Since

$$Q/A = \frac{k_{TS}}{k_{TS}} \Delta T_{TS} = \frac{k_{MK}}{k_{MK}} \Delta T_{MK}$$

Then

$$k_{TS} = \frac{\ell_{TS}}{\ell_{MK}} \frac{\Delta T_{MK}}{\Delta T_{TS}} k_{MK}$$

The conductivities $\mathbf{k}_{\mbox{MK}}$ and $\mathbf{k}_{\mbox{TS}}$ are evaluated at the arithmetic mean temperatures,

$$T_{MK} = T_{MK}(HOT SIDE) - \frac{\Delta^{T}_{MK}}{2}$$

and
 $T_{TS} = T_{TS}(HOT SIDE) - \frac{\Delta^{T}_{TS}}{2}$

It is noted there is very little difference between analytical and measured K's thereby indicating that the analytical model is very good.

The through panel fastener, Figure 3, was designed for low heat transfer by ensuring that the three modes of heat transfer were minimized. block radiation and restrict air convection, fibrous insulation, DYNAFLEX®, was placed within the cavity of the fastener. In that way DYNAFLEX's very low thermal conductivity is substituted for those two terms. Therefore, the heat transfer becomes primarily a conduction problem. Metal conduction was minimized by keeping the cross-sectional area (the conducting area) normal to the panel axis small, i.e., fastener conduction area/panel total area is a small value. The maximum number of fasteners per panel is four. So, for a panel that is 304.8 mm by 304.88 mm, the conduction area ratio is four times each fastener conduction area/(304.8 by 304.8). This is (4π) (14.478) (0.127)/(304.8by 304.8) = 0.00025. The effective thermal conductivity of a panel with fasteners, k_{TWP} , may be approximated by $k_{TWP} = 0.00025 k_p + (1 - 10.00025 k_p)$ $0.00025)k_T$ where k_p is fastener material conductivity and k_T is panel thermal conductivity.

The CERACHROME® contribution is not included because its conductivity is nearly the same as $\mathbf{k}_{\mathsf{T}}.$

This equation may be rewritten as

$$(k_{TWP}/k_T) = 0.00025 (k_p/k_T) + 1 - 0.00025$$

At 900F (482.2C), $k_p = 11.92$ Btu/hr ft f(20.6228 w/mk) and from Figure 32 $k_T = 0.07$ Btu/hr ft F (0.1211 w/mk)

Thus

$$(k_{TWP}/k_T) = 0.00025 (11.92/0.07) + 1 - 0.00025$$

 $(k_{TWP}/k_T) = 1.04$

i.e., a maximum increase of 4.0 percent would be expected for the panels' k.

Based on these test results, the thermal conductivities used in the thermal math model are considered acceptable for future thermal analyses.

7/ STRUCTURAL PERFORMANCE

7.1 GENERAL

The purpose of the structural evaluation program was twofold:

- a. To provide basic mechanical properties of the brazed INCONEL 617 sandwich.
- b. To predict and verify the structural performance of the panel design and manufacturing processes.
- 7.2 MECHANICAL PROPERTIES OF INCONEL 617 HONEYCOMB SANDWICH
 The basic mechanical property testing was performed on coupon size
 specimens while the structural and thermal performance verification was
 performed on a full size panel. The full size panel tests verify that
 the panel is able to withstand a realistic simultaneous pressure load and
 temperature environment. The coupon test quantifies the strength
 properties of the material system and verifies that the panel met all of
 the design requirements. An outline of the test program with the number
 of specimens involved is provided in Table 11.

During the coupon testing, face sheets and sandwich structures with various gages (including the final design configuration) were tested. Specimens were ultrasonically C-scanned prior to testing. Specimen locations were marked on the C-scans and the panels. Photographs were taken of the panels for a permanent record of their location. Each specimen was identified by a number/letter combination that related it to the panel from which it came and to the type of test that was performed on it.

The remainder of this section provides details of all of the testing. These details include a description of:

- a. Test specimen configuration
- b. Test apparatus and procedures
- c. Test results.
- 7.2.1 FACE SHEET TENSION TESTS -- Tests were conducted to determine the basic mechanical properties of INCONEL 617 foil material as received and after being subjected to various conditions. These conditions included:
 - a. Processed/brazed to honeycomb core
 - b. Pretest exposure to 1,366°K (2,000°F) for either 5 or 25 hours.

Test temperatures varied from room temperature to 1,366°K (2,000°F). The following mechanical properties were determined:

- a. Yield (Fty) and ultimate (Ftu) stress
- b. Percentage elongation (e)
- c. Modulus of elasticity (E).

The modulus of elasticity values were measured from load - deflection curves that were plotted in conjunction with a linear variable differential transformer (LVDT) and with the Instron test machine.

The specimens, except for the as-received specimens, were cut from brazed INCONEL 617 honeycomb sandwich panels. The honeycomb core was removed from the face sheets with a high speed grinder. The overall specimen size was 2 inches by 10 inches with a 1-inch wide test section. Two thicknesses were tested: 0.076 mm (0.003 inch) and 0.127 mm (0.005 inch).

The test program and results are summarized in Figure 33 and in Tables 12, 13, and 14. The groupings are by duration of pretest 1,366°K (2,000°F) exposure. These are respectively: none, 5 hours and 25 hours. The pretest thermal exposure was performed to determine the degradation of material properties over the life of a panel. It has been estimated that these panels would be exposed to 1,256° to 1,366°K (1,800° to 2,000°F) environment for approximately 300 seconds during every flight or approximately 8 hours for 100 flights. This duration was conservatively bracketed by the 5 and 25 hour exposure times and using the upper temperature value of 1,366°K (2,000°F). The atmosphere used for this exposure was sea level air -- a conservative condition since most entry heating occurs at a high altitude.

The test specimens were separated from the core prior to exposure. The effects of this pretest exposure are discussed in subsequent paragraphs and are also illustrated metallographically in Figures 34 through 36. Figure 35 shows the typical microstructure of the INCONEL 617 alloy in the solution-annealed condition. Figure 35 shows the foil after being brazed to honeycomb core and being exposed for 5 hours at $1,366^{\circ}K$ ($2,000^{\circ}F$). The rough upper surface is braze alloy. A very thin gray layer on the surfaces indicates an oxidation film. Dark lines and spots indicate the beginnings of intergranular oxidation. Figure 36 is the same except the exposure duration of $1,366^{\circ}K$ ($2,000^{\circ}F$) temperature has been increased to 25 hours. The oxidation film has increased in thickness and the intergranular oxidation is significantly greater.

Table 12 summarizes the testing on specimens that had not been subjected to pretest thermal exposure. The yield and ultimate strength values are comparable to, or slightly higher than, published data. The percent elongation of the as-received material is considerably lower (12 percent versus 31 percent) than the values on the material certification sheets that were produced by the material vendor. Subsequent investigations showed that if the test specimens are more carefully prepared (to ensure failures in the two-inch test area) and load rates are reduced to 0.51 mm (0.02 inch) per minute crosshead speed, elongation values increase from 12 percent to 34 percent.

Tables 13 and 14 summarize the testing on specimens with 5 and 25 hours of 1,366°K (2,000°F) pretest thermal exposure, respectively. These exposures have only a moderate impact on the yield strength values. However, the ultimate strength and the percent elongation values continue to decrease with the duration of pretest exposure. The reduction stabilizes at 1,366°K (2,000°F) and the number of hours of exposure does not affect these test values. Therefore, the pretest exposure durations are most critical for room temperature mechanical properties. Figures 37 and 38 display this point graphically. Percent elongation values show the same trend.

7.2.2 CREEP TESTS -- Tests were conducted to determine the long term strength of INCONEL 617 foil material when exposed to elevated temperature and sustained stress levels. The test matrix is shown in Table 15. As shown, the testing included temperatures from 1,089° to 1,366°K (1,500° to 2,000°F) and foil conditions of as-received and processed/brazed-to-honeycomb core.

The initial test specimen configuration was 6.4 mm (1/4 inch) wide by 51 mm (2 inches) long by 0.08 mm (0.003 inch) thick. This specimen proved to be adequate at the $1,089^{\circ}K$ ($1,500^{\circ}F$) test temperature condition; however at $1,366^{\circ}K$ ($2,000^{\circ}F$) it produced widely scattered results which are not reported. The reason for the scatter is believed

to be related to the small size of the test specimen and the resulting small load requirements. The specimen size was then substantially increased to 19 mm by 76 mm by 0.08 mm (3/4 inch by 3 inches by 0.003 inch) for all 1,255°K and 1,366°K (1,800°F and 2,000°F) testing. The test setup is shown schematically in Figure 39 with an overall photograph in Figure 40. Note that specimens are dead weight loaded and that creep deflections are automatically plotted as a function of time. The larger creep specimen, along with three thermocouple probes, is shown in Figure 41.

The test data is tabulated in Table 16 and is shown in the Larson-Miller plots in Figure 42. The total elevated temperature life of the structure is estimated to be 8 hours for 100 missions (See Section 7.2.1) with a maximum temperature of $1,311^{\circ}K$ ($1,900^{\circ}F$). For the purpose of comparing this test data with actual stress-temperature conditions, specific stress and temperature points are provided. This comparison conservatively treats the total eight hours as occurring at each temperature point examined.

7.2.3 EDGEWISE COMPRESSION TESTS

These tests were conducted to evaluate the ability of thin foil gages to carry significant compressive loads. These thin gages, when bonded into sandwich structure, do have some initial waviness. Therefore, it had been theorized that these sheets were already in a buckled condition and as such would be unable to carry any significant compressive loads. The tests completely disproved this theory because ultimate compressive stresses of considerable magnitude were measured.

The test specimens were brazed INCONEL sandwich with a square cell core that had a height of approximately 7.1 mm (0.280 inch). The specimens were 76 mm (3 inches) wide and 89 mm (3.5 inches) long in the direction of the applied load. The ends of the specimens were potted with an acrylic compound to provide local support and uniform load application. The specimens with 0.08 mm (0.003-inch) thick face sheets had considerably more initial face sheet waviness than those with 0.13 mm (0.005 inch) face sheets.

The test program and results are tabulated in Table 17. The specimens were tested in accordance with ASTM C364. The failure mode for all specimens was intracell buckling, which is to be expected for a sandwich with thin face sheets and large ratios of cell size to face sheet thickness. Figure 43 plots the test results versus analytical results. This figure shows that there is close agreement for the 0.13 mm (0.005-inch) face sheets. However, the test results for the 0.08 mm (0.003-inch) face sheets are approximately 30 to 75 percent higher than the analytical results.

The analytical results are from an intracell buckling equation (See equation C12.5.1 of Reference 18) which was developed from tests of standard sandwich specimens and almost certainly never involved these foil type gages. Therefore, the discrepancy between analytical and test results is attributable to inaccuracy in the analytical method when dealing with large ratios of cell size to face sheet thickness and large ratios of braze alloy to face sheet thickness. Consequently, the analysis conservatively underestimated the specimen load carrying capability.

7.2.4 FLATWISE TENSION TESTS

Flatwise tension testing is a standard method of assessing the process procedures of the bonding operation. The results are not directly used in the stress analysis but they do provide a means of comparing the strength of various bonded joints. The data presented includes room and elevated temperature data on both environmentally exposed and unexposed specimens. The environmental exposure was in a 1,366°K (2,000°F) air furnace for either 5 or 25 hours. The test setup for room temperature testing is shown in Figure 44. The test setup for elevated temperature testing is shown in Figure 45.

The test plan is shown in Tables 18A and 18B. As shown, some of the E panel specimens received a pretest thermal exposure. As in the case of the face sheet tension tests, this exposure was performed to determine

the degradation of material properties over the life of a panel. It has been estimated that these panels would be exposed to 1,256° to 1,366°K (1,800° to 2,000°F) environment for approximately 300 seconds during every flight or a total of approximately 8 hours for the 100-flight design life. This duration was conservatively bracketed by the 5 and 25 hour exposure times at 2000°F. There were several conservative procedures used during this pretest exposure. They include:

- a. A 1,366°K (2,000°F) exposure (the upper temperature limit)
- b. A test atmosphere of sea level air (actual exposure will be at elevations where there is rarefied atmosphere)
- c. Exposing the separate 76 mm by 76 mm (3-inch by 3-inch) specimens rather than an entire panel with edge closures which would protect the interior part of the panel.

Another feature of the test program was room temperature and elevated temperature testing. The room temperature specimens had loading blocks adhesively bonded to them. The elevated temperature specimens had the loading blocks brazed to them using 1.978-0.02C -13.13Cr -3.4 Fe braze alloy at 2175°F. This process did not interfere with the sandwich brazed joints.

The first panels that were fabricated for these tests were designated AFT and GFT. As defined in Tables 18A and 18B, they had $0.08~\rm mm$ ($0.003~\rm inch$) face sheets and $4.6~\rm mm$ ($0.1875~\rm inch$) cell core. C-scans of these panels showed varying degrees of bond quality. In order to correlate C-scan readings with joint strength, the panels were cut into specimens and tested. Test specimen numbers and results were recorded on the C-scans. As a result, a high degree of correlation was identified between the C-scans and the flatwise tension strengths. Those specimens that showed low quality bonds in the C-scans had considerably less strength (on an average of $1/3~\rm to$ 1/2) than those without disbonds. The disbonds in these panels were attributed to early development problems in the manufacturing process. The test results for these panels are not

included in this report. The E panel, which was fabricated subsequent to the A and G panels, had ideal C-scans. The configuration of the E panel is identical to the production panel design (0.005 each face sheets and 0.25 inch cell core). Only the results of E panel flatwise tension tests are reported here.

The room temperature test results for the E panel specimens are tabulated in Table 19 and plotted graphically in Figure 46. The reduction in strength after exposure to 1,366°K (2,000°F) is attributed to oxidation of the core and not to oxidation of the braze joint. This conclusion is supported by the failure modes and photomicrographs of the joints. In fact, some of the 25 hour exposed core had failed locally prior to loading due to the exposure. The failure mode for the unexposed specimens was 100 percent in the brazed joint while the exposed specimens had large areas of core failure. Figures 47 through 49 show the brazed joint of a core cell wall and a face sheet after various amounts of 1,366°K (2,000°F) exposure. It is evident that the cell wall is being attacked much more severely than the braze joint. It should be noted that even the core in the center of the specimens was oxidized. The air passageways are through the cell nodes which are spotwelded together.

The elevated temperature results of testing specimens from the E panel require a special explanation. The low results shown in Table 19 are the result of extenuating circumstances. After 25 hours at 1,366°K (2,000°F), the 76.2 mm by 76.2 mm (3-inch by 3-inch) specimens were severely warped as well as oxidized. This warpage could have been alleviated by subjecting a large panel to the exposure instead of the small 76.2 mm by 76.2 mm (3-inch by 3-inch) specimens. For room temperature tests, this warpage does not cause any great problems because additional adhesive can be added to fill gaps between the loading blocks and face sheet. However, the elevated temperature specimens require that loading blocks be attached by brazing alloy, which can not fill large gaps like the adhesive. Consequently, during the test, there was uneven loading and local separation of the face sheets from the loading blocks. These

test conditions and results must be considered unrealistic to any actual service operation.

As stated previously, flatwise tension results by themselves are not a normal part of the stress analysis. However, they provide a means to evaluate the effects of other parameters on joint integrity. In this test it has been shown that a 1,366°K (2,000°F) exposure in an oxygen rich atmosphere over a period of time has a significantly deleterious effect on the sandwich structure. However, the conservative nature of the testing has measured reductions that far exceed those which would result from the flight design life.

7.3 STRUCTURAL ANALYSIS OF FULL-SIZE PANEL

7.3.1 FINITE ELEMENT MODEL -- A finite element model of the entire panel was constructed in order to determine the internal stresses and external deflections for the design conditions discussed in Section 3.0. The model, shown in Figure 50, has approximately 390 nodes. The coded model input sample is shown in the appendix. The computer program selected for the analysis was NASTRAN. The selection was based on the fact that this program has industry wide acceptance and use, and Rohr has extensive experience with it. The upper INCONEL sandwich and the lower titanium sandwich panels were modeled using one inch by one inch panel elements which are defined as CQUAD4. CQUAD4 panel elements are special plate members that represent sandwich structure. The sidewalls were modeled as a combination of two different elements. These elements are CSHEAR, to represent the sidewalls capability to react shear loads, and pinned CBAR members, to represent the beam-column load capability of the corrugated flutes. The clip and bayonet attach fittings are modeled as rod members as shown in Section A-A and B-B in Figure 50. Rods were selected so there would not be any bending capability in these supports. In addition, the rods were given an axial stiffness which was determined from a full panel pull test. Subsequently, the pressure and thermal gradients described in Section 3.0 were applied to the model. The stress levels are discussed below and the deflection values are discussed in Section 7.4 of this report.

7.3.2 STATIC STRESS ANALYSIS -- The calculated stresses, for the two ascent conditions and the one descent condition, are shown on Figures 51-A through 51-F. These stresses are superimposed on finite element models in order to provide a representation of the stress distributions within the panels. The panels have two center lines of symmetry, therefore only a quarter of the panel is required to define the internal stress distributions. The stresses shown on the INCONEL and titanium honeycomb are principal major or minor stresses with (+) representing tension and (-) compression. The stresses shown on the sidewalls in parentheses are shear stresses and the other sidewall stresses are axial loads in the bars representing the corrugations.

The all positive margins of safety for the critical stresses from these conditions are tabulated in Tables 21A and 21B. Included in these tables are allowables for the INCONEL and titanium honeycomb, INCONEL sidewalls and the titanium bayonet attach fittings. The critical failure mode for the honeycomb structure is intracell buckling. The allowable curve for the INCONEL is shown in Figure 52. It is based on room temperature test data from edgewise compression tests (Table 17), and temperature reduction factors based on modulus of elasticity (E). The E values were generated during the mechanical property testing and are averages of specimens pretest exposed to 5 hours of 2000°F and those exposed to 25 hours of 2000°F (see Figure 38). The titanium honeycomb allowable, shown in Figure 53, is based on equations in Reference 17. The INCONEL sidewall, which was found not to be stability critical, has allowables based on $F_{\mbox{t}\,\mbox{v}}$ shown in Figure 37 . The value used is an average between the curves for 5-hour and 25-hour pretest exposure of 2000°F. The titanium bayonet fittings have allowables based on MIL-HDBK-5D values for Ti-6A1-4V.

In conclusion, the successful structural panel testing verifies the analysis and the integrity of the panel.

7.4 THERMAL/PRESSURE TESTS ON FULL-SIZE PANEL

- 7.4.1 GENERAL -- In order to verify the structural integrity of a total panel assembly, a series of thermal and pressure gradient tests were conducted. A panel assembly, which was fabricated to Rohr Engineering Drawing 195-256, was installed in a test fixture in a manner which accurately simulated installation to a vehicle surface. The test panel, which was instrumented with thermocouples and dial indicators, is shown in Figure 28.
- 7.4.2 TEST FIXTURE AND INSTRUMENTATION The test fixture (Rohr Drawing 501-560) is shown schematically in Figure 54. Photographs of the test fixture and instrumentation are shown in Figures 55 through 59. In the schematic, starting at the bottom, there are dial indicators with ceramic dowels which penetrate through the quartz lamps. The quartz lamp bank array is shown in Figure 56. The ceramic dowels, shown protruding through the lamps, must penetrate a water chamber which circulates water to cool and protect the aluminum support plate. Surrounding the lamp bank (not shown in Figure 54 but shown in Figure 55) is a rectangular, gold-plated reflecting shield which keeps the heat in and on the panel.

A completely independent and separate assembly is suspended above the lamp assembly. This assembly contains:

- a. The test panel
- b. Mounting clips
- c. Seals
- d. A pressure chamber to load the panel.

The test panel has its exterior surface exposed directly to the lamp array. The panel is clipped into the base of the pressure chamber. Figure 57 shows this chamber in an inverted position and without the cover plate. Note that the clips and bayonet fittings for the normal mating structure are included.

Also shown in this figure and in the schematic of Figure 54 are two different seals. The design and function of these silicone seals is very important. The seal on the outer perimeter simulates the NOMEX® pad that would be installed on the shuttle vehicle. This pad is compressed during panel installation and provides a tight fit for the panel. It also reacts crushing pressure loads that push the panel against the vehicle. The test seal is purposely not bonded to the panel so that it will not inadvertently react blowoff pressure loads that pull the panel away from the vehicle.

The inner seal is referred to as the flap seal. It provides the seal to the pressure chamber. As such it must be bonded to the panel but also must not react any blowoff loads. This is possible because of the seal design. The seal is L-shaped and, since it is made from silicone rubber, does not have any bending stiffness. Consequently, the seal is incapable of reacting load and therefore all loads go through the clips as they should. Figures 58 and 59 show views of this seal as it attaches to the bottom of the panel. Also note the holes in the panel. The holes assure that all pressure gradients will be across the outer INCONEL sandwich structure. These holes are not part of the panel design but are incorporated in the test to accommodate rapid pressure changes that could take place during the test but not during actual flight conditions. The final part of the fixture is a cover plate which is bolted on. A vacuum pump provides crush pressure and an external air supply provides blowoff pressure. Both are monitored by a pressure gage.

Figure 55 shows, on the far left, a Thermac Controller (Research Incorporated) which regulates power to the quartz lamps. To the right of this is a Fluke Data Logger which records the temperatures from the thermocouples.

NOMEX is a registered trademark of E. I. duPont de Nemours & Co., Inc., Wilmington, Delaware.

7.4.3 TEST PROGRAM AND RESULTS -- The testing was performed according to the six conditions outlined in Table 22. The intent of the program was to cover as many possible design conditions as practical and to do so in a conservative manner. Note that the critical design conditions described in Section 3.0 are met or exceeded by these test conditions. The ascent design conditions listed in Table 1 is exceeded by test conditions V and VI and the descent design condition is approximated by test Condition IV. At the time of the test, the precise temperature gradient had not been calculated. In this test program, the design ultimate burst and crush pressures were initially applied at room temperature. Subsequently the panel was subjected to the maximum 1.366°K/477°K (2.000°F/400°F) temperature gradient without pressure loads. Next, the design ultimate burst pressure load was applied in combination with a conservative temperature gradient (a higher temperature gradient than that expected in combination with pressure) of 812°K/311°K (1,000°F/100°F). After successfully passing this severe condition, the loading was increased to determine the margin of safety. At 25 KPa (3.6 psi) an air leak occurred at two of the corners of the panel and the testing was terminated. Other than these small holes at the two corners there was no discernible damage to the panel.

The panel was repaired by placing a 0.08 mm (0.003 inch) thick piece of INCONEL 617 foil over the holes. Resistance spot welds were then made between the foil and the panel to close the holes. The panel was reinstalled in the test fixture, heated to the 812°K (1,000°F)/311°K (100°F) temperature gradient, and pressurized to 25 KPa (3.6 psi) at which time a pressure drop was again noted. The panel was removed and evaluated. A failure in the INCONEL 617 side closures at the titanium 6Al-4V intersection as shown in Figures 60 and 61 was noted. Tack welds used to stabilize the panel during LID bonding held when the bonded area between tack welds separated causing small tears in the side wall. Since the failure was primarily in the base material, no other attempt was made to repair the panel.

The heat-up rates on the test panel were controlled and were those calculated for an entry condition for body point 1300. These temperatures were monitored during heat-up and during load application. Table 23 shows the temperatures for various burst pressure loads. The results verify the consistent and uniform temperature gradients that were established throughout the panel.

Figure 62 plots the deflections at the center of the top surface of the panel versus applied pressure loads. For the severe test condition of 14 KPa (2 psi) burst pressure plus 811°K/311°K (1,000°F/100°F) temperature gradient, the deflection at the center of the panel was 4.0 mm (0.156 inch): 1.5 mm (0.060 inch) due to thermal and 2.4 mm (0.096 inch) due to pressure. In order to relate this to panel bow, Figure 63 was plotted. The plot shows deflection values at all 4 corners of the panel, the middle of one side and the center of the panel for the severe condition. The plotted deflections are those due to pressure only and the thermal deflections are presented in tabular form. In order to calculate maximum panel bow (an aerodynamics performance concern), the value of the corner with the smallest deflection is subtracted from the panel center deflection. For the 14 KPa (2 psi) plus 811°K/311°K (1,000°F/100°F) condition, corner number one had the smallest deflection. This value was 1.3 mm (0.051 inch): 0.3 mm (0.011 inch) due to thermal and 1.0 mm (0.040 inch) due to pressure. Therefore, the maximum panel bow for the ultimate design condition was 2.7 mm (0.105 inch). The nonlinearity in the deflection curves above the 14 KPa (2 psi) load is attributed to bending in the clips.

Table 20 presents a comparison of deflections obtained from the test versus those calculated by the NASTRAN finite element model described in Section 7.3. As shown, the analytical procedure underestimated the test results except for the 2 psi room temperature blowoff condition. These higher analytical results are surmised to be from an under prediction of the stiffness of the bayonet support fittings.

In conclusion, the panel design and manufacturing processes were demonstrated by full scale tests to be completely adequate to withstand the design criteria defined in Table 1. Consideration should be given as to whether protective coatings are necessary for the exterior of these panels in order to reduce the oxidation effects of elevated temperatures.

8/ CONCLUSIONS

A metallic reusable Thermal Protection System (TPS) panel with the potential for withstanding $1,366\,^{\circ}$ K ($2,000\,^{\circ}$ F) was designed to protect areas of space reentry vehicles where the temperature does not exceed $1,311\,^{\circ}$ K ($1,900\,^{\circ}$ F) and the pressure load is no greater than 14 KPa (2 PSI).

Test panels were fabricated using existing production facilities and processes. It was demonstrated that the panels can be mass produced by processing large quantities of parts simultaneously. One array of twenty panels and five extra panels were fabricated and delivered to NASA Langley Research Center for additional Testing. A TPS panel was designed using the thermal requirements for Space Shuttle body point 1300 as representative design criteria. Thermal and structural analyses were performed. Component specimens and full size panels were fabricated and tested to verify the design. Comparison of analytical and test data substantiate the analysis methods and verify the thermal and structural performance of the panels.

9/ REFERENCES

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Table 1. Design Criteria - Body Point 1300

	OU.	TER PANEL		INNER PANEL		
LOAD	ΔP (PSI) ULTIMATE	T OUTER SURFACE °F	ΔT °F	ΔP (PSI) ULTIMATE	T INSIDE SURFACE °F	∆T °F
Ascent	± 2.0 ^a	650 ^b	50 ^C	0.0 ^d	100 ^C	0c
Descent	0.0 ^e	1900 ^f	31 ^f	0.0 ^c	208 ^f	24 ^f

- a) This value is a contractual requirement and can be either a blowoff or crush pressure.
- b) Reference 7, Pages 2-9.
- c) Assumes same heating rates for ascent as descent. See Figure 4.
- d) Panel has vent holes through inner sandwich layer.
- e) Reference 7 reports that descent shock pressures occur only after panel has cooled down to nearly ambient temperature.
- f) Maximum temperature and thermal gradient for BP/1300. See Figure 4.

Table 2. Trajectory 14414.1C

TIME SECONDS	BODY POINT SURFACE °F	
0	250	
100	650	
200	1,100	
300	1,700	
400	1,800	
450	1,900	
500	1,900	
600	1,870	
700	1,800	
800	1,630	
900	1,530	
1,000	1,420	
1,100	1,280	
1,200	1,120	
1,300	1,000	
1,400	1,050	
1,500	650	
1,600	280	
1,700	120	
2,000*	120	

The typical surface temperature history for body point 1300 of space shuttle trajectory 14414.1C (once around).

* Extended time to ensure no continued temperature rise of the aluminum structure.

Table 3. Typical Time Pressure History for Shuttle Body Point 1300

TIME SECONDS	BODY POINT 1300 PRESSURE LBF/SQ FT
0	0.01087
100	0.08373
200	1.01035
350	17.69237
450	26.51259
550	32.69217
600	37.28221
650	40.55208
700	44.23282
750	43.79233
800	42.45522
850	43.84506
1,050	59.55977
1,150	69.60561
1,200	74.90744
1,250	69.32410
1,300	68.84630
1,350	61.44383
1,400	71.89258
1,450	66.87845
1,500	76.15733
1,550	91.65157
1,600	115.08743
1,650	171.99934
1,750	2116.217
2,000*	2116.217

^{*} Extended time to ensure no continued temperature rise of the aluminum structure.

Table 4. Candidate Materials for Thermal Protection System (1 to 10 listed in order of 1,366 $^\circ$ K (2,000 $^\circ$ F) 100 hour Creep Rupture Specific Strengths)

	MAIN ALLOY CONSTITUENT	Nickel	Nickel	Nickel	Iron	Nickel	Cobalt		Cobalt	Nickel	Nickel	Nickel	Iron	Nickel	Nickel
1,366°K (2,000°F)/100 HR CREEP RUPTURE	Ξ.		5.46 (21.9)	_	_	2.22 (8.9)		•	_	_	_	1.37 (5.5)	$\overline{}$	$\overline{}$	0.9 (4.0)
) E N	g/cm ³ (1b/in ³)	_	8.86 (0.320)	_	_	8.36 (0.302)	9.13 (0.330)		_	8.44 (0.305)	8.53 (0.308)	8.05 (0.291)	_	8.41 (0.304)	8.22 (0.297)
SHORT TIME U.T.S. Mpa (ksi)	1,255°K (1,800°F)	276 (40)	110 (16)	_	_	_	$\overline{}$		_	_	_	(6) 29	_	_	_
SHORT TIN	1,366°K (2,000°F)	ι.	90 (13)	_	_	_	_		_	_	_	34 (5)	_	N.A.	90 (13)
R CREEP STRENGTH (ksi)	1,255°K (1,800°F)		(6.6) 99	$\overline{}$						_	_	23 (3.4)	_	_	
100 HR CREEP RUPTURE STRENGTH Mpa (ksi)	1,366°K (2,000°F)	N.A.	_	55 (8)	_	19 (2.7)			15 (2.2)	12 (1.8)	12 (1.7)	11 (1.6)	11 (1.6)	J	8 (1.2)
	MATERIAL	Rene 41	T.D. Ni	T.D. Ni-Cr	INCOLOY 802	INCONEL 617	=) S097	Haynes 25)	Haynes 188	INCONEL 625	Nimonic 86	INCONEL 601	Haynes 556	INCONEL 600	HASTELLOY X
		A	8	ပ —	-	-2	m	,	4	2	9		∞	6	10

a - 1,323°K b - 1,173°K

N.A. - Not Available A - Precipitation Hardened Superalloy B-C - Oxide Dispersed Alloys 1-10 - Solid Solution Strengthened Superalloys

Table 5. Thermophysical Properties of INCONEL 617 Honeycomb (4-20 Core) Thickness 0.293 Inch

T	Cp
(°F)	(BTU/LB-°F)
78.	0.100
200.	0.104
400.	0.111
600.	0.117
800.	0.124
1000.	0.131
1200.	0.137
1400.	0.144
1600.	0.150
2000.	0.163

Т 	k* (BTU/FT-HR-°F)
100.	0.1482
200.	0.1666
400.	0.2041
600.	0.2512
800.	0.3107
1000.	0.3846
1200.	0.4755
1400.	0.5858
1600.	0.7178
1800.	0.8738
2000.	1.0565

* Effective Thermal Conductivity calculated by standard methods (see Reference 15)

INCONEL Density =
$$521.0 \text{ lbs/ft}^3$$
 $\epsilon \text{ external} = 0.80 \quad \epsilon \text{ internal} = 0.60$

Table 6. Thermophysical Properties of DYNAFLEX

(°F)	Cp (BTU/LB-°F)
240	0.202
440	0.233
640	0.252
840	0.267
1040	0.274
1240	0.280
1640	0.284

					k			
Ţ	Р			(BT	U/HR-FT-	°F)		
<u>°F</u>	PSF	0.0278	0.2785	2.785	27.85	139.2	278.4	2116
200		0.0043	0.0048	0.0088	0.0178	0.0206	0.0211	0.0215
400		0.0106	0.0111	0.0150	0.0261	0.0306	0.0313	0.0320
600)	0.0173	0.0177	0.0214	0.0342	0.0403	0.0414	0.0425
800)	0.0255	0.0259	0.0294	0.0433	0.0512	0.0327	0.0542
1000)	0.0369	0.0373	0.0406	0.0553	0.0649	0.0669	0.0688
1200)	0.0530	0.0534	0.0566	0.0718	0.0830	0.0854	0.0879
1400)	0.0706	0.0710	0.0740	0.0896	0.1023	0.1052	0.1083
1600	}	0.0930	0.0933	0.0962	0.1119	0.1262	0.1296	0.1333
1800)	0.1156	0.1159	0.1187	0.1345	0.1501	0.1540	0.1583
2000)	0.1466	0.1469	0.1496	0.1654	0.1823	0.1867	0.1917
2200)	0.1827	0.1829	0.1855	0.2013	0.2193	0.2243	0.2300
2400)	0.2173	0.2175	0.2200	0.2358	0.2549	0.2606	0.2670

Density = 6.0 lbs/ft^3

Reference: Manufacturer Brochure for 2116 PSF Values. The k values at other pressures are estimated by methods of References 16 and 17.

Table 7. Thermophysical Properties of Q FIBER FELT

	Cp (BTU/LB-°F)
240	0.202
440	0.233
840	0.267
1040	0.274
1240	0.280
1640	0.2845

					k			
T	Р			(BT	U/HR-FT-	°F)		
°F	PSF	0.0278	0.2785	2.785	27.85	139.2	278.4	2116
			,					
10	0	0.0020	0.0030	0.0085	0.0155	0.0168	0.0170	0.0172
20	0	0.0050	0.0059	0.0116	0.0201	0.0220	0.0223	0.0225
30	0	0.0078	0.0087	0.0145	0.0244	0.0268	0.0272	0.0275
40	0	0.0107	0.0116	0.0174	0.0285	0.0316	0.0321	0.0325
60	0	0.0181	0.0188	0.0246	0.0380	0.0425	0.0432	0.0438
80	0	0.0267	0.0274	0.0330	0.0483	0.0541	0.0551	0.0560
100	0	0.0364	0.0370	0.0425	0.0592	0.0665	0.0678	0.0690
120	0	0.0476	0.0483	0.0534	0.0713	0.0802	0.0818	0.0833
140	0	0.0624	0.0635	0.0685	0.0876	0.0981	0.1001	0.1020
150	0	0.0765	0.0770	0.0820	0.1015	0.1127	0.1149	0.1170

Density = 3.5 lb/ft^3

Reference: Manufacturer Brochure for 2116 PSF Values. The k values at other pressures are estimated by methods of References 16 and 17.

Table 8. Thermophysical Properties of Titanium Honeycomb (3-20 Core) Thickness 0.185 Inch

T	Cp
(°F)	(BTU/LB-°F)
0.	0.140
200.	0.140
400.	0.145
600.	0.148
800.	0.155
1000.	0.166

T °F	k* (BTU/FT-HR-°F)
0.	0.0651
100.	0.0764
200.	0.0883
400.	0.1133
600.	0.1413
800.	0.1754
1000.	0.2149

* Effective Thermal Conductivity calculated by standard methods (see Reference 15)

Titanium Density =
$$281.5 \text{ lb/ft}^3$$

 ε external = 0.80 ε internal = 0.18

**Aluminum properties used were:

density =
$$169 \text{ lb/ft}^3$$

C_p = $0.229 \text{ BTU/lb} - \text{°F}$

Table 9. Thermophysical Properties of MIN-K

Т 	k (BTU/FT-HR-°F)
100.	0.0145
200.	0.0148
300.	0.0153
400.	0.0159
500.	0.0166
1200.	0.0225

Reference: Manufacturers Brochure

Table 10. Boundary Temperatures Used and Comparison of Predicted and Measured Steady State Temperatures

MEASU BOUND TEMPER		T13 _T	T13 _A ANALYTICAL	PERCENT ERROR $(T13_A-T13_T) \times 100$	Q _A * BTU/HR ANALYTICAL	** Q _T BTU/HR
^T 1	T ₁₇	MEASURED	PREDICT ION	T13 _T	PREDICT ION	TEST
			CENT EF	R OF TILE		
305.5	90.3	190.7	189.2	-0.78	17.38	17.61
605.8		385.2	391.7	1.68	48.6	47.31
1	154.9	610.0	618.2	1.34	88.65	86.25
1189.2	201.8	866.2	871.8	0.65	137.2	134.58
1474.9	265.9	1139.8	1136.3	-0.31	191.88	192.00
1799.4	354.5	1461.5	1445.9	- 1.07	258.91	266.18
			ADJACENT	TO SIDEWALL		
305.5	90.3	198.8	203.7	2.46	19.93	19.05
605.8	123.6	397.6	416.5	4.75	53.44	49.65
896.2	154.9	615.1	647.5	5.27	94.89	87.31
1189.2	201.8	877.0	902.0	2.85	144.38	137.17
1474.9	265.9	1155.5	1165.5	0.86	199.61	196.16
1799.4	354.5	1476.0	1472.3	-0.25	266.10	270.50

^{*} Based on predicted temperature at T_{13}

^{**} Based on measured temperature at T_{13}

Table 11. Structural Test Summary

TEST TYPE	PRETEST ELEVATED TEMP. EXPOSURE	ELEVATED TEMP. TESTS	NUMBER OF SPECIMENS
Face Sheet Tension ¹	Yes	Yes	61
Face Sheet Creep ¹	No	Yes	16
Edgewise Compression ²	No	No	35
Flatwise Tension ²	Yes	Yes	30
Full Panel Pressure/Temperature Gradient	No	Yes	1

 $^{^{\}rm 1}$ Test specimens are INCONEL foil material both as-received and after process/brazing

² Test specimens are INCONEL brazed sandwich

Table 12. INCONEL 617 Face Sheet Tension Tests - No Pretest Environmental Exposure - Average Values

$\frac{E^1}{\text{GPA}}$ (KSI × 10 ³)	221. (32.0)	195.5 (28.3)	184 (26.7)	98.6 (14.3)	77.2 (11.2)	$\frac{110.0}{(16.1)}$	21.0 (3.05)
o 26	12.1	34.3	2.7	5.5	10.7	22.3	3.0
Ftul MPA (KSI)	889 (129.0)	1007 (146.1)	836 (121.3)	468 (67.9)	219 (31.7)	219 (31.7)	62 (9.0)
Ftyl MPA (KSI)			478 (69.3)	226 (32.8)	118 (17.1)	139 (20.1)	45 (6.5)
TEST TEMPERATURE K (°F)	Room Temp.	Room Temp	Room Temp.	1089 (1500)	1255 (1800)	1255 (1800)	1366 (2000)
NUMBER OF SPECIMENS	ĸ	S	S	ß	ις	4	S.
SPECIMEN THICKNESS mm (inch)	0.076 (0.003)	0.076 (0.003)	0.076 (0.003)	0.076 (0.003)	0.076 (0.003)	0.127 (0.005)	0.076 (0.003)
SPECIMEN CONFIGURATION	A-1 As-Received	A-2 As Received ²	Processed/Brazed to Honeycomb Core				
SPE	A-1	A-2	8.	ပ	D.	ů.	г.

1 Average values.

 $^{^2}$ This is a repeat of A-1 tests with specimens more carefully prepared and load rates reduced to $0.51\,\mathrm{mm}$ (0.02 inches) per minute crosshead speed.

Table 13. INCONEL 617 Face Sheet Tension Tests - 5 Hours of 2,000°F Pretest Exposure - Average Values

E GPA (KSI x 10 ³)	219.	72.4 (10.5)	77.9 (11.3)	19.2 (2.78)
o 26	1.6	2.3	8.3	1.5
Ftu MPA (KSI)	585. (84.9)	148. (21.5)	174. (25.3)	39.
Fty MPA (KSI)	412. (59.8)	127. (18.4)	137. (19.8)	37. (5.3)
TEST TEMPERATURE K (°F)	Room Temp.	1255 (1800)	1255 (1800)	1366 (2000)
NUMBER OF SPECIMENS	2	5	S	2
SPECIMEN THICKNESS mm (inch)	0.076 (0.003)	0.076 (0.003)	0.127 (0.005)	0.076 (0.003)
SPECIMEN CONFIGURATION	Processed/Brazed to Honeycomb Core	Processed/Brazed to Honeycomb Core	Processed/Brazed to Honeycomb Core	Processed/Brazed to Honeycomb Core
SPE	й	Ξ.	ij	ب

Table 14. INCONEL 617 Face Sheet Tension Tests - 25 Hours of 2,000°F Pretest Exposure Average Values

E GPA (KSI x 10 ³)	216.0 (31.4)	32.0 (4.64)
a 26	9.0	1.2
Ftu MPA (KSI)	404.0 (58.6)	46.0 (6.7)
Fty MPA (KSI)	388.0 (56.2)	41.0 (6.0)
TEST TEMPERATURE K (°F)	Room Temp.	1366 (2000)
NUMBER OF SPECIMENS	2	2
SPECIMEN THICKNESS mm (inch)	0.076 (0.003)	0.076 (0.003)
SPECIMEN CONFIGURATION	K. Processed/Brazed to Honeycomb Core	Processed/Brazed to Honeycomb Core
SPE	×.	۲.

Table 15. Creep Tests - INCONEL 617 (0.003 Inch Foil)

SPECIMEN CONFIGURATION	SPECIMEN SIZE (inch)	TEST TEMPERATURE	NUMBER OF SPECIMENS
Sheet Material As- Received	1/4 x 2 x 0.003	1500°F	6
Processed/Brazed to Honeycomb Core	1/4 x 2 x 0.003	1500°F	5
Processed/Brazed to Honeycomb Core	3/4 x 3 x 0.003	1800°F	2
Processed/Brazed to Honeycomb Core	3/4 x 3 x 0.003	2000°F	<u>3</u> (a)
TOTAL			16

⁽a) Some other tests were conducted at this temperature with the smaller (1/4 inch x 2 inch) test specimens. However, the test results were considered invalid and are not reported.

Table 16. INCONEL 617 Creep Rupture Test Results

	SPECIMEN NO.	STRESS (KSI)	TEMPERATURE (°F)	TIME (HRS.)	$P = (460 + T) (20 + log t)10^{-3}$
AS REC'D MATERIAL	1A 2A 3A 4A 5A	20.0 20.0 17.0 13.0 15.0	1500 1500 1500 1500 1500	20.0 19.5 29.9 167.3 ² 107.1 44.5	41.8 41.7 42.1 43.6 43.2 42.4
тсн	1C 2C 3C 4C 5C	17.0 21.0 21.0 21.0 20.0	1500 1500 1500 1500 1500	116.2 ² 32.0 63.6 69.8 98.3	43.2 42.2 42.7 42.8 43.1
BRAZED SANDWICH FACE SHEET	N1 1 N2 1 N4 1 N5 1 N6 1	1.0 1.0 6.0 6.0 2.0	2000 2000 1800 1800 2000	90.0 330.4 46.8 53.7 92.3	54.0 55.4 49.7 49.1 54.0

¹ These specimens have 3/4 inch wide test area while others have 1/4 inch wide test area.

2 No Failure

P = Larson-Miller Parameter

T = Temperature, °F

t = Time, hrs.

Table 17. Edgewise Compression - INCONEL 617 Sandwich (See Figure 43)

SPECIMEN CONFIGURATION	NUMBER OF SPECIMENS	Fcu ¹ (KSI)
0.003 inch Face Sheet, 3/16 inch with 0.002 inch Core	S	57.0
0.003 inch Face Sheet, 1/4 inch with 0.002 inch Core	5	43.8
0.003 inch Face Sheet, 3/8 inch with 0.002 inch Core	2	28.3
0.003 inch Face Sheet, 3/16 inch with 0.0015 inch Core	5	53.0
0.005 inch Face Sheet, 3/16 inch with 0.002 inch Core	5	60.1
0.005 inch Face Sheet, 1/4 inch with 0.002 inch Core	5	52.5
0.005 inch Face Sheet, 3/8 inch with 0.002 inch Core	12	39.3
TOTAL	35	

1 These are average values

2 Testing was performed at room temperature

Table 18A. Explanation of Panel Identification Number

XFT-Y

 \underline{X} - Panel Configuration

FT - Type of Test - Flatwise Tension

 \underline{Y} - Test Conditions

TEST CONDITIONS

Y	Pre-Test Environment	Test Temperature	Material Processing Status
А	None	Room Temperature	As Received
В	None	Room Temperature	Brazed
С	None	1500°F	Brazed
D	None	2000°F	Brazed
E	5 Hours @ 2000°F air furnace	Room Temperature	Brazed
F	5 Hours @ 2000°F air furnace	2000°F	Brazed
G	25 Hours @ 2000°F air furnace	Room Temperature	Brazed
Н	25 Hours @ 2000°F air furnace	2000°F	Brazed

PANEL CONFIGURATION

Table 18B

Х	Face Sheet Thickness (Inches)	Core Cell Size (Inches)	Core Foil Thickness (Inches)
Д*	0.003	3/16	0.002
В	0.003	1/4	0.002
С	0.003	3/8	0.002
D	0.005	3/16	0.002
E*	0.005	1/4	0.002
F	0.005	3/8	0.002
G*	0.003	3/16	0.0015

 $[\]star$ Only these configurations were selected for flatwise tension testing

Table 19. Flatwise Tension Tests (3-Inch by 3-Inch One-Layer INCONEL 617 Honeycomb Core Sandwich)

AVERAGE FLATWISE TENSION STRENGTH (PSI	C-Scans of many of the specimens from panels AFT and GFT indicated defective braze joints. Test results are not reported.				1682	799	213	18 2		CORE FOIL THICKNESS (INCHES)	25	315	25
νl	Note:									CORE FOIL	0.002	0.0015	0.002
NUMBER OF SPECIMENS	10	10	LG	5	10	5	10	5	09				
TEST TEMPERATURE	Room Temperature	Room Temperature	1,500°F	2,000°F	Room Temperature	Room Temperature	Room Temperature	2,000°F		CORE CELL SIZE (INCHES)	3/16	3/16	1/4
PRE-TEST ENVIRONMENT						5 Hours @ 2,000°F Air Furnace R	25 Hours @ 2,000°F Air Furnace R	25 Hours @ 2,000°F Air Furnace 2		FACE SHEET THICKNESS (INCHES)	0.003	0.003	0.005
PANEL IDENTIFICATION 1 PRE-	AFT - B None	GFT - B None	GFT - C None	GFT - D None	EFT - B None	EFT - E 5 H	EFT - G 25 H	EFT - H 25 H	TOTAL	PANEL CONFIGURATION	A	ၒၟ	ш

See Table 18A for explanation. These specimens were severely warped by the 25-hour furnace exposure and consequently the brazed loading blocks cause highly concentrated loads. **--** 2

Table 20. Panel Deflections - Analytical versus Test

		LOCATION							
L	OADING CONDITION	CENTER OF	PANEL 1	MIDDLE O		CORNER O	F PANEL ³		
Α.	2 psi Blowoff - Room Temperature								
	Analytical Test	0.147 0.098		0.117 0.078		0.120 0.044			
В.	2 psi Crush Room Temperature				, , , , , , , , , , , , , , , , , , ,				
	Analytical Test	-0.032 -0.039		0.001 0.017		0.005 0.008			
c.	Max. Thermal Gradient								
	Analytical (1900°F)(203°F) Test (2000°F)(400°F)	0.093 0.138		0.055		0.016 0.049			

¹⁾ Grid #1120 (see Figure 50); Dial Indicator #3 (see Figure 65)

²⁾ Grid #1126 (see Figure 50); Dial Indicator #5 (see Figure 65)

³⁾ Grid #616 (see Figure 50); Dial Indicator #1 (see Figure 65)

Table 21A. Ascent Conditions (Ref. Table 1)
Stress Levels and Margins of Safety

	ASCENT (CONDITION	ASCENT		MINIMUM	
COMPONENT	STRESS (CRUSH) (PSI)	STRESS (BLOWOFF) (PSI)	ALLOWABLE (PSI)	FAILURE MODE	MARGIN OF SAFETY	
INCONEL Sandwich	-9,700 ^a	-11,500a	41,000 ^b	Intracell Buckling	+2.57	
INCONEL Sidewall	-7,690 ^C	5,540 ^c	48,700 ^d	Axial and Shear	+5.33	
Titanium Sandwich	-2,900 ^e	-14,900 ^f	24 , 000 ⁹	Intracell Buckling	+0.61	
Titanium Clip	0	180,500 ^h	183,300 ⁱ	Bending	+0.02	
Titanium Clip	0	120,300 ^j	140,400 ^k	Bending	+0.17	

- (+) Tension
- (-) Compression
- a) QUAD4 Element ID 1021, principal stress
- b) Intracell Buckling Allowable @ T = 600°F Ref. Figure 52
- c) Shear 103, Bars 103 and 104 (shear stress and average stress in two adjacent bars), principal stress
- d) Not critical in stability, use F_{ty} at T = 400°F. Reference Figure 37 (average of 5-hour and 25-hour data.
- e) QUAD4 109 (t = 0.006 inch) principal stress
- f) QUAD4 210 (t = 0.003 inch) principal stress
- q) Intracell Buckling Allowable @ T = 100°F. Reference Figure 53.
- h) 2.0 PSI ultimate + temperature environment
- i) Ultimate plastic bending allowable + temperature
- j) 1.33 psi limit + temperature environment
- k) Limit plastic bending allowables + temperature environment.

Table 21B. Descent Conditions (Ref. Table 1)
Stress Levels and Margins of Safety

COMPONENT	DESCENT CONDITION STRESS	DESCENT ALLOWABLE	FAILURE MODE	MINIMUM MARGIN OF SAFETY
INCONEL Sandwich	-9,100 ^a	14,500 ^b	Intracell Buckling	+0.59
INCONEL Sidewall	-5,200 ^c	33,500 ^d	Axial and Shear	+5.44
Titanium Sandwich	-9,100 ^e	23 , 200 ^f	Intracell Buckling	+1.55
Titanium Clip	0	111,400 ⁹		High

- (+) Tension
- (-) Compression
- a) Intracell Buckling Allowable @ T = 1869°F (Maximum stress is on inner surface of INCO Honeycomb). Reference Figure 52.
- b) Shear Panel 12, Bar 324 and Rod 325 (shear stress and average axial stresses in bar and rod), principal stress.
- c) Not critical in stability, use F_{ty} @ T = 1050°F. Reference Figure 37 (average of 5-hour and 25-hour data.
- d) QUAD4 210 (t = 0.003 inch).
- e) Intracell Buckling Allowable @ T = 200°F Reference Figure 53.
- f) Ultimate Plastic Bending Allowable + Temperature.

Table 22. INCONEL Bi-Metal Panel Pressure-Thermal Gradient Test

Condition I	Room Temperature	2 psi (crush)
Condition II	1,000°F/100°F	2 psi (crush)
Condition III	Room Temperature	2 psi (burst)
Condition IV	2,000°F/400°F	-0- psi
Condition V	1,000°F/100°F	2 psi (burst)
Condition VI	1,000°F/100°F	3.6 psi (burst) Panel Air Leak

8 Thermocouples 6 Dial Indicators Controlled Heatup Rates

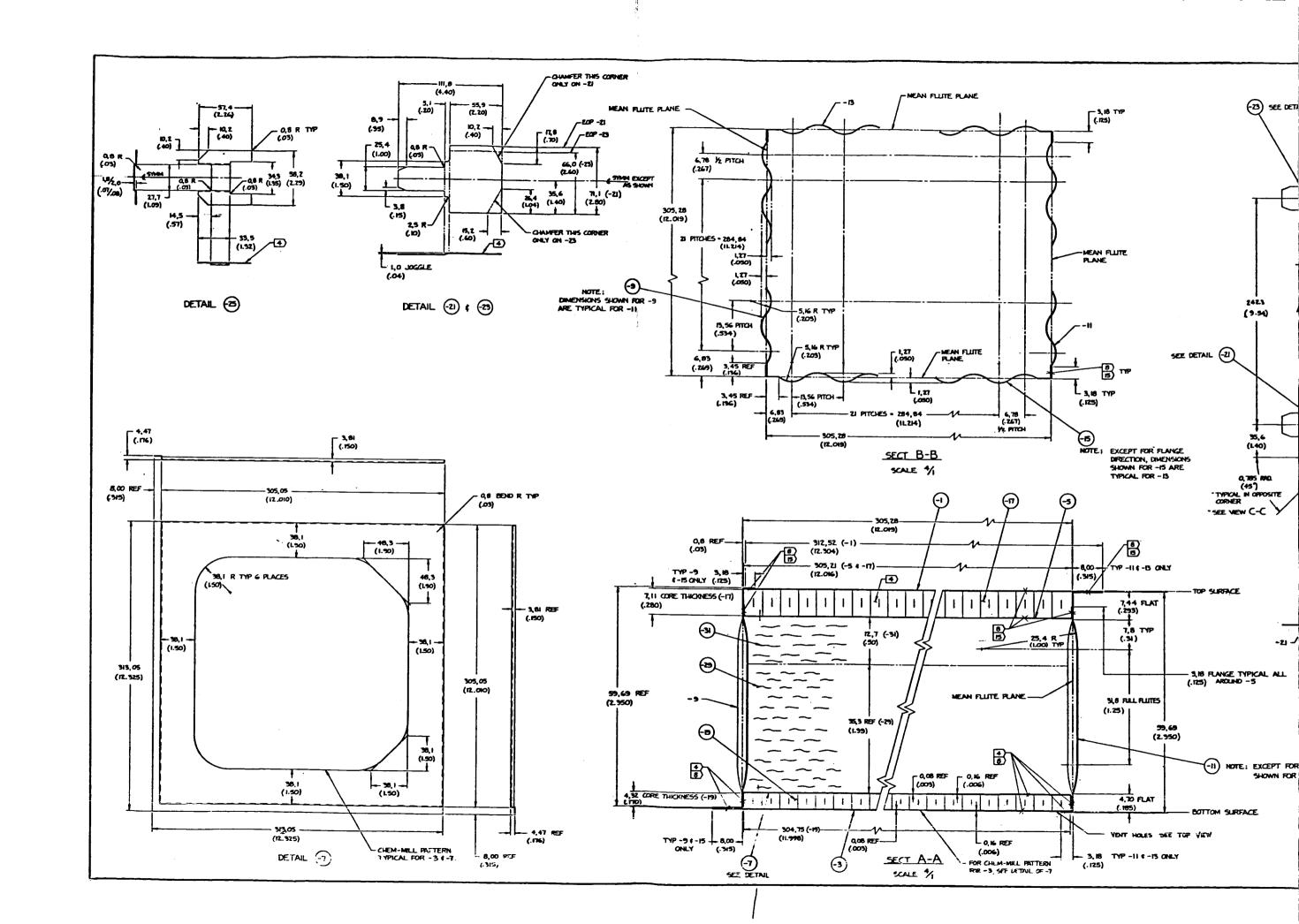
Table 23. Test Panel Temperature Profile During Burst Pressure Test

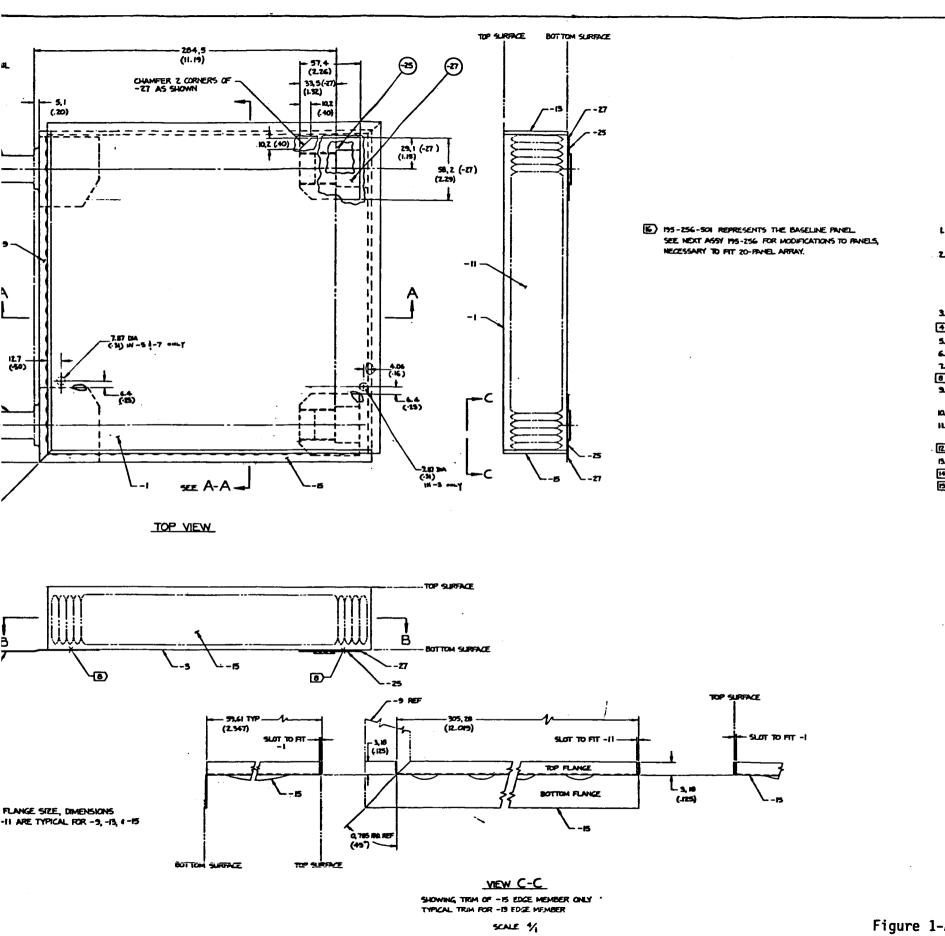
APPLIED PRESSURE LOAD			THERMOC	OUPLE REA	ADING °K	(°F)		
KPa (PSI)	1	2	3	4	5	6	7	8
6.89	814.8	815.9	413	332	336	814.8	816.5	811.5
(+1.00)	(1007)	(1009)	(283)	(137)	(145)	(1007)	(1010)	(1001)
13.8	812.6	814.8	414	328	333	812.6	813.7	808
(+2.00)	(1003)	(1007)	(285)	(131)	(139)	(1003)	(1005)	(995)
17.2	813.2	822.0	416	323	330	813.7	818.2	817.6
(+2.50)	(1004)	(1020)	(289)	(121)	(134)	(1005)	(1013)	(1012)
20.7	793	805	435	319	333	793	800	813.2
(+3.00)	(967)	(989)	(323)	(114)	(139)	(968)	(980)	(1004)
24.1	785	796	452	312	323	784	780	810
(+3.50)	(949)	(974)	(354)	(101)	(122)	(952)	(945)	(998)
24.8	834.3	855.9	435	310	312	837.0	855.4	820.9
(+3.60)	(1043)	(1081)	(324)	(93)	(102)	(1047)	(1080)	(1018)

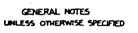
Thermocouples 1,2, 6, 7 and 8 are on the hot outer surface of the panel.

Thermocouple 3 is on the side of the panel.

Thermocouples 4 and 5 are on the panel inner surface.







- L DIMENSIONS IN S.I. LINITS & (CLISTOMARY LINITS); mm (in.), rad (deq), N/m1 (torr), K ($^{\circ}$ F)
- Z. TOLERANCES:

LINEAR	X 12	.x • # .7	.xx • ± .25
	l.± • K.)	.XX - 1 .03	xxx - ± .010)
ANGLILAR	1. ± • X.	IO. E - XX.	100. ± • xxx.
	(x ·±z	.x - ± .5	XX +± 05)

- 3. SEE 195-255 FOR TOOL TO MAKE CLOSURES, -9, -11, -15, 4-15
- 4) LID PLATE THESE SURFACES, & ENTIRE N/C CORE PER RPS (1.89), TYPE B IN PREPARATION FOR BONDING
- 5. FORM TITANIUM PARTS PER RPS 14.02
- 6. PROCESS -17 (-19 CORES PER RPS 11.86-9
- 7. CHEM-MILL TITANIUM PER RPS H.10. TOLERANCES ± 0,03 (± 001). VACILIM DECASSING MANDATORY
- 5 SPOTWELD COMPONENTS IN POSITION IN PREPARATION FOR BONDING
- 9. IDENTIFY ALL PARTS & ASSYS PER RPS 19.99. RUBBER STAMP LOCATION OPTIONAL IMPRESSION STAMPING NOT PERMITTED.
- IO. PROCESS BOND ASSY PER RPS II.68
- IL BOND IN A VACULUM FURNACE AT A PRESSURE OF .0066 N/m² (5 ×10 -6 torr) \$ TEMPERATURE OF 1214 K (1725°F).
- (2) OXIDIZE THIS SURFACE ONLY E-8
- 15. PERMISSIBLE WAVINESS ± 0,81 (± .032)
- (4) CORE MATERIAL PER RMS 110
- 5 SUPERALLOY L.I.D. BOND PER RPS (TBD)

\sqcup		-31	CERACHROME. 36.1	2 Kg/m³ (6	.o us/FT*)		
Ц	%	-25	Q-FIBER FELT 56.0	77 Kg/m³ (3.	5 (/FT)		
Ц	\perp	 	<u> </u>	<u> </u>			\Box
\vdash	- -					ļ	₩.
\sqcup	2	- 27	DOUBLER	TI-GAL-4V	.020 × 3 × 3	AMS 49(1	
\sqcup	2	- 25	CLIP		.020 × 3 × 3		
Ш	11	~ Z3	TONGLE		.032 × 3 × 5		LT
	1	- 2 .1	TONGLE	TI-CAL-4V	.032 × 3 × 5	AMS 4711	
	1	-19	CORE 3-20-RCB	TI-3AL-25V	.170 × 13 × 13	1	
\Box	11	-17	CORE 4-20-RB-P	INCO 617	.280 × 13 × 13	(H)	
П		-15	CLOSURE	1	.003 × 4 × 14		1
	1	-13			1		
П	11	-11					
П		-9	CLOSURE	INCO 617	.003 × 4 × 14		
\prod	11	-7	SEPTUM - LOWER	TI-GAL-4V	.006 = 14 = 14	AMS 4911	
\coprod	1	-5	SEPTUM - UPPER	INCO GIT	.005 × 13 × 13	T	
		-3	SKIN - LOWER	TI-GAL-4V	.006 × 13 × 13	AH5 4917	
П	1	-1	SKIN-UPPER	INCO GIT	.005 × 13 × 13		
Ш	$\bot \bot$						
Ш	$\bot \bot$		<u> </u>	L			$\Box \Gamma$
Ц			ļ		l <u>. </u>		
Ц		- 501	PANEL ASSY (6)				
TY RE	G D	PART NO.	DESCRIPTION	MATERIAL	SIZE (IL)	SPEC	нт

.

Figure 1-A. Panel Assambly

INDUSTRIES, INC.

PANEL ASSY-

51583 195-756

NASA T.P.S.

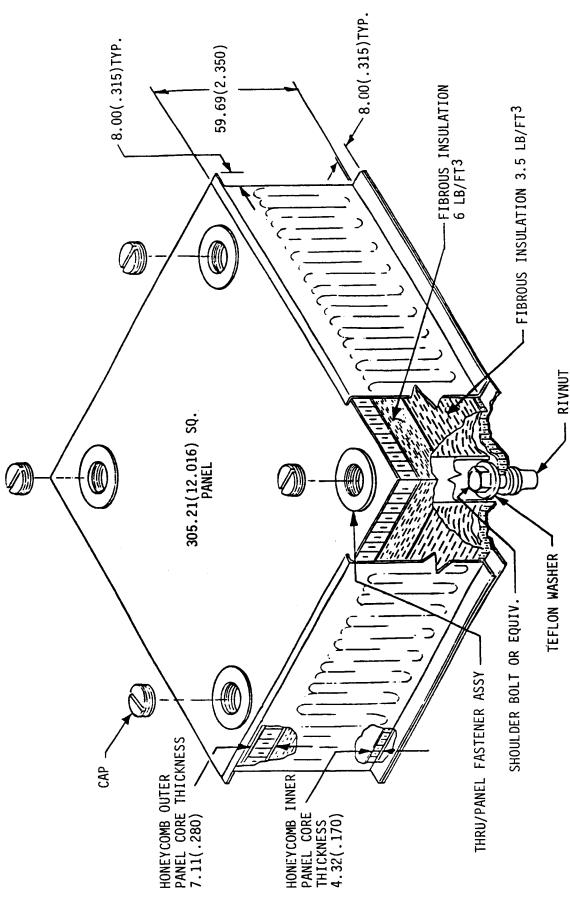


Figure 1-B. Schematic of Bi-Metal Silica Sandwich Panel

Figure 2. Bayonet Attachment Scheme

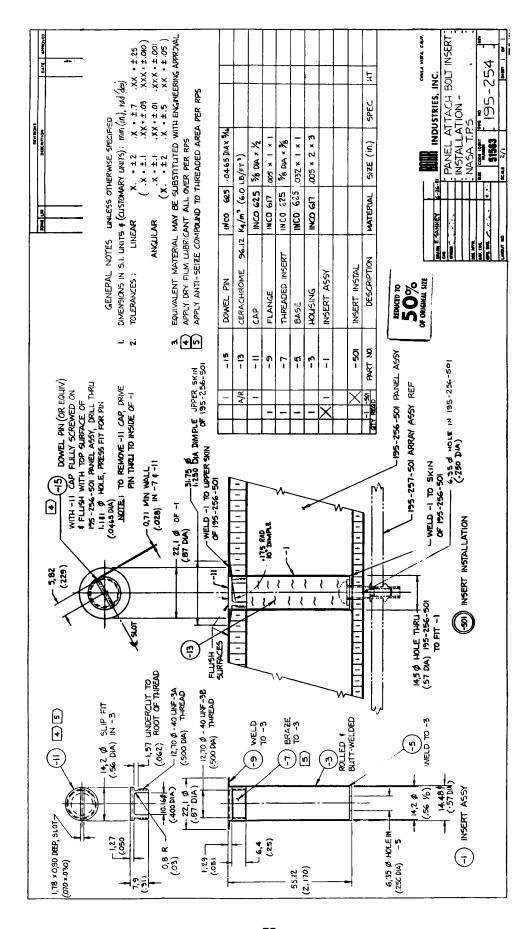


Figure 3. Panel Attach Bolt Insert Installation

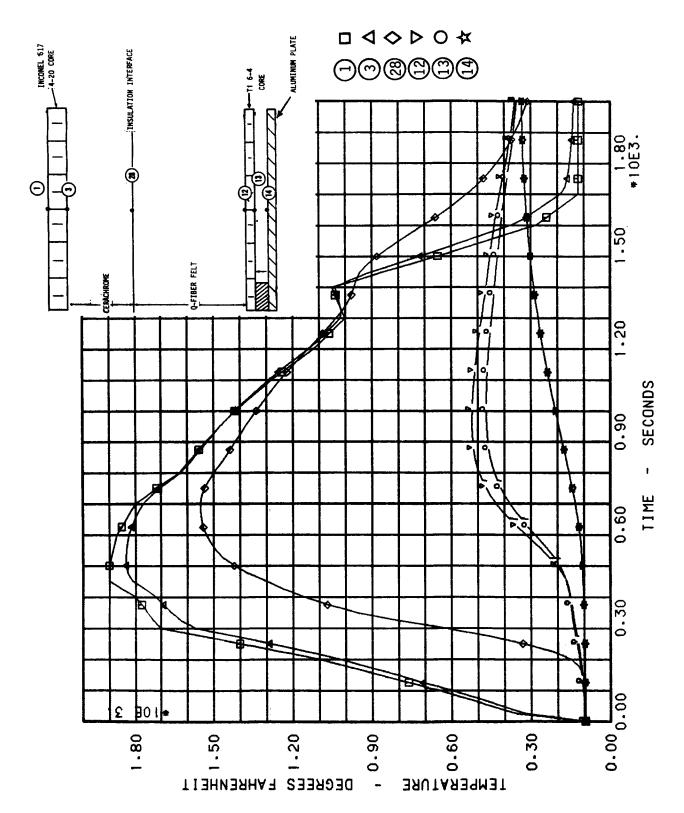


FIGURE 4 TYPICAL RESULTS FOR TRANSIENT ANALYSIS

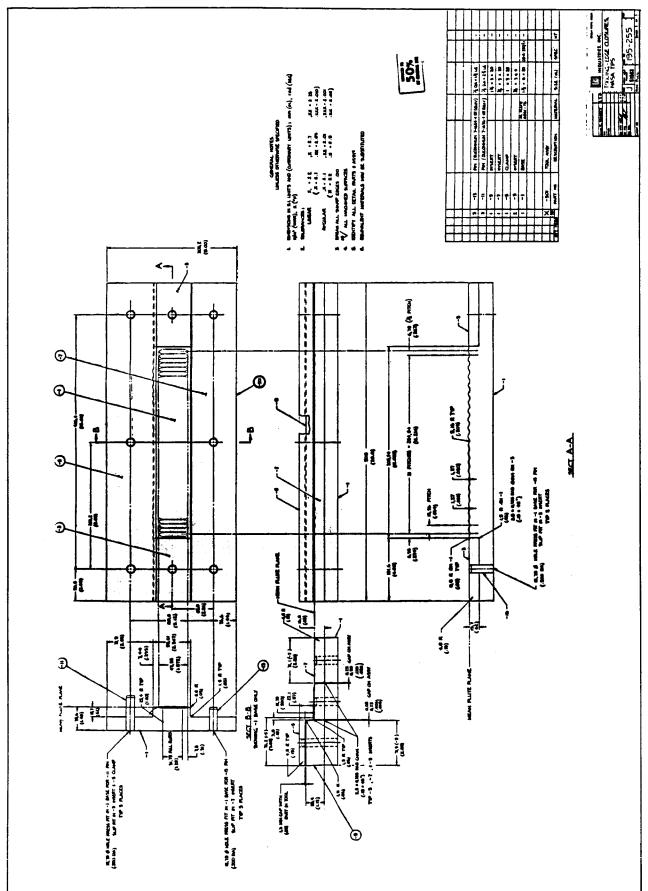


Figure 5. Edge Closure Tool

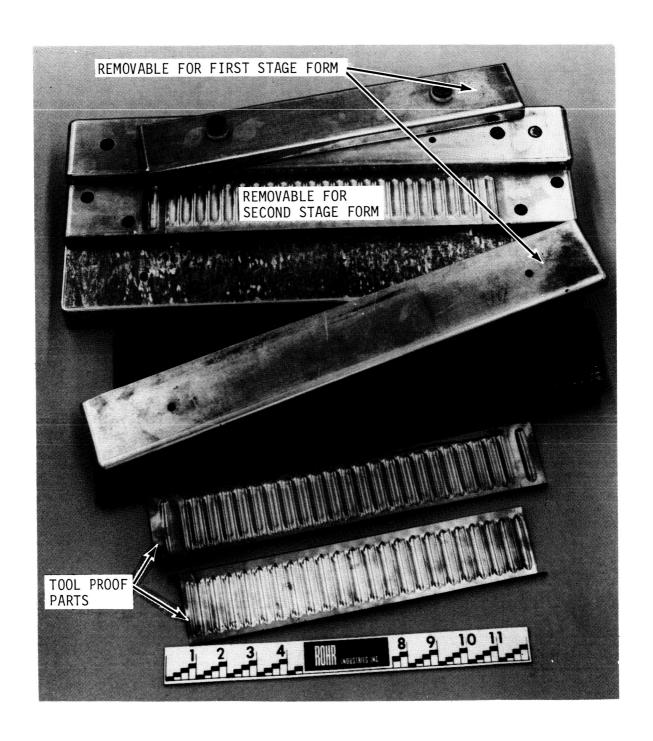


Figure 6. 6061 Aluminum Form Block With Removable Details and Tool Proof Parts

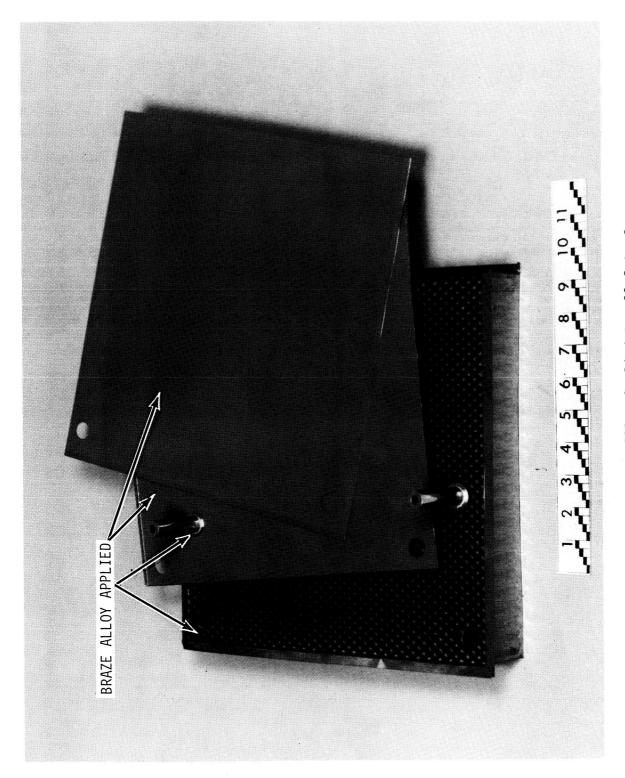


Figure 7. Braze Alloy Applied to all Interfaces

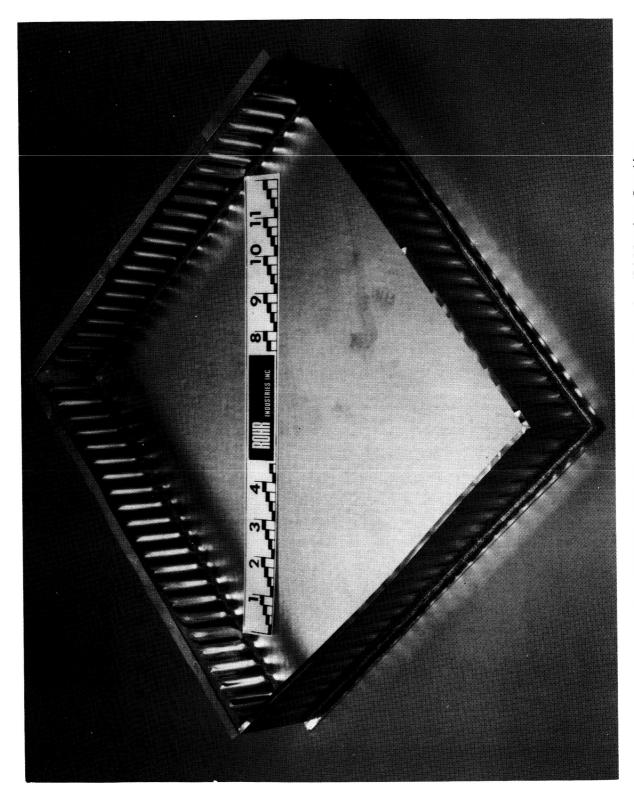


Figure 8. INCONEL 617 Subassembly after Brazing/Diffusion Bonding

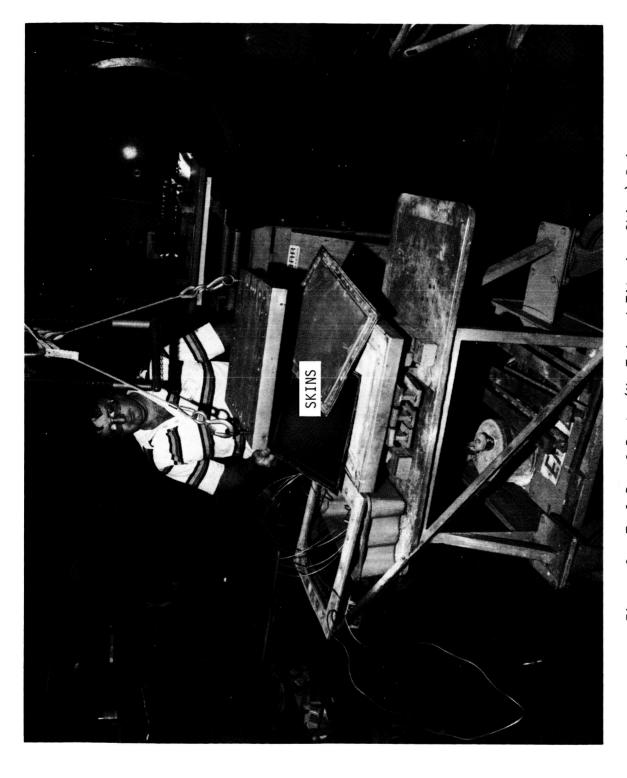


Figure 9. Tool Proof Parts (Un-Trimmed Titanium Skins) Being Removed from SPF Tool

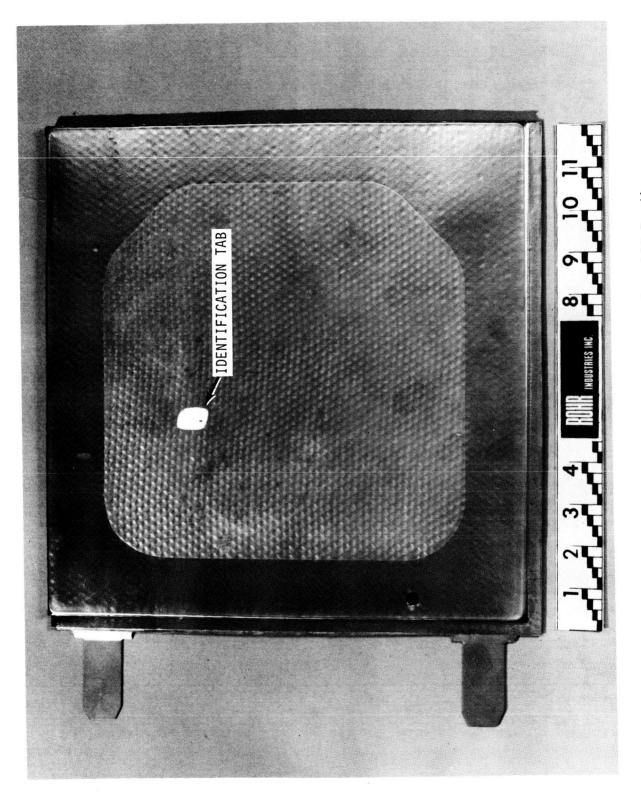


Figure 10. Top of Titanium Subassembly after LID Bonding

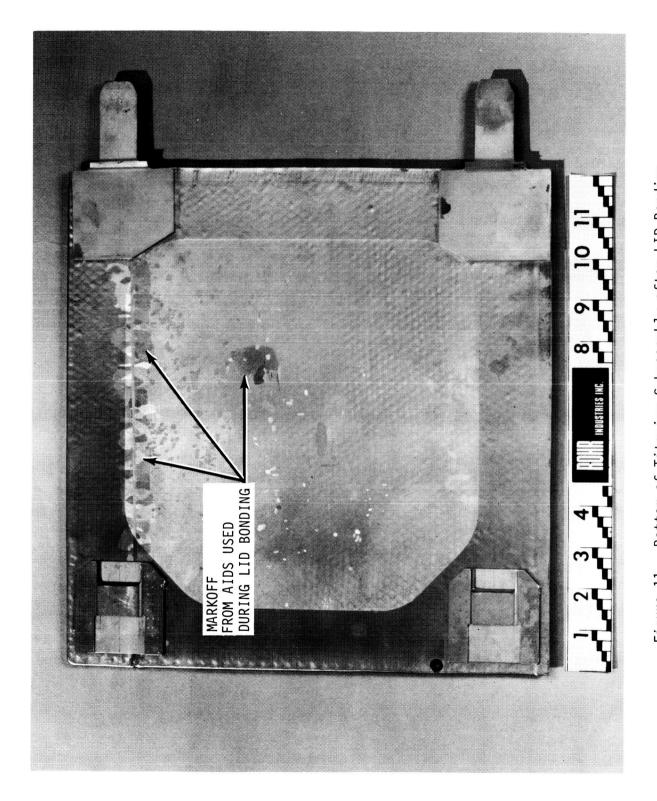


Figure 11. Bottom of Titanium Subassembly after LID Bonding

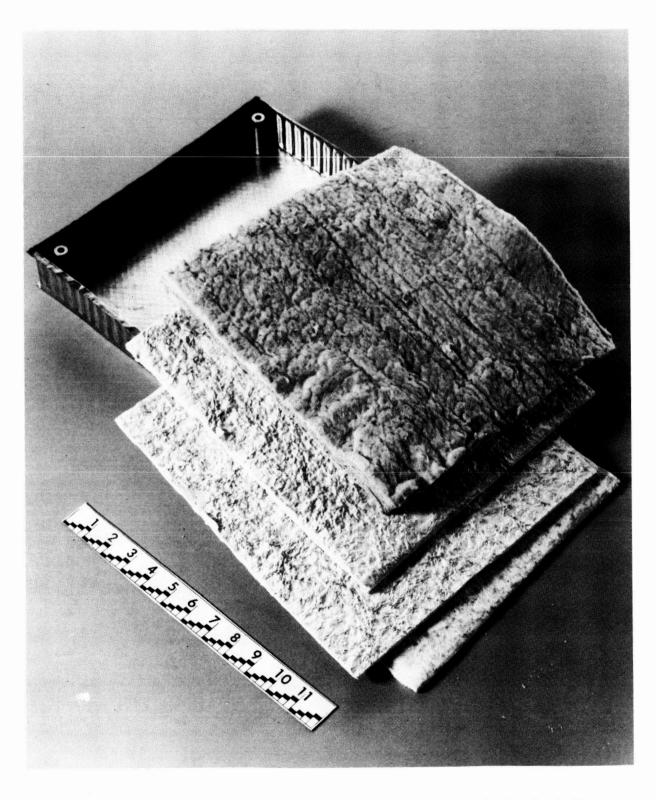


Figure 12. Pre-cut Q-Fiber Felt and DYNAFLEX Ready to be Installed into the INCONEL 617 Subassembly

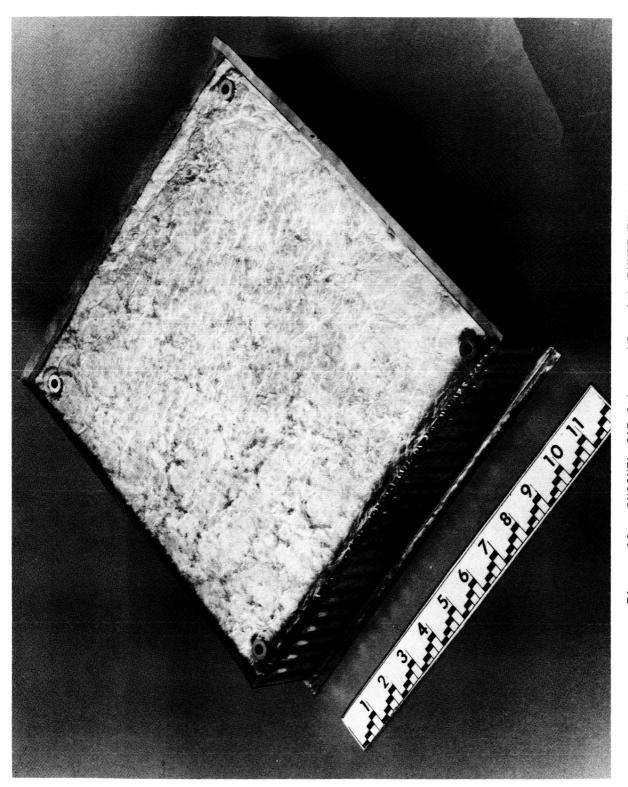


Figure 13. INCONEL 617 Subassembly with DYNAFLEX and Q-FIBER FELT Installed



Figure 14. Bi-Metal Assembly Being Laid up for LID Bonding

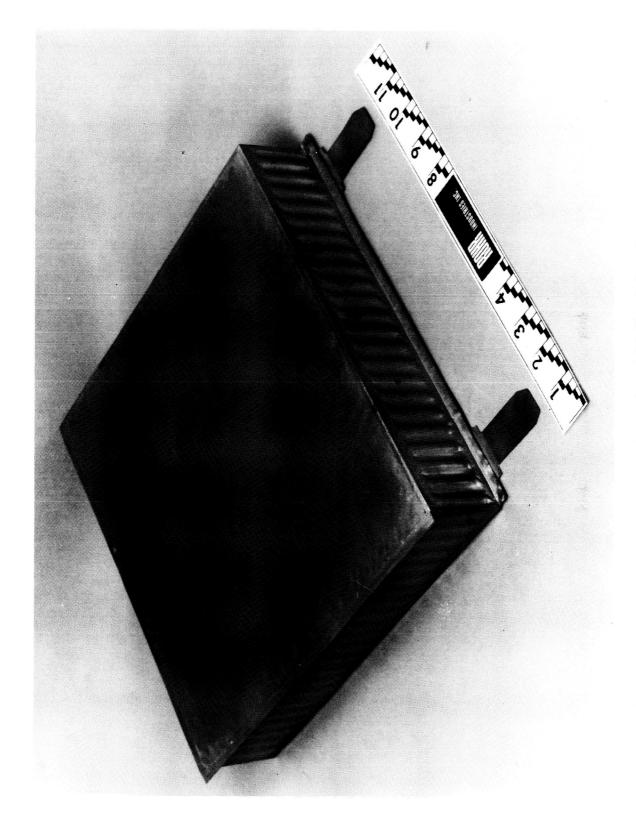


Figure 15. Top of Completed Bi-Metal Panel

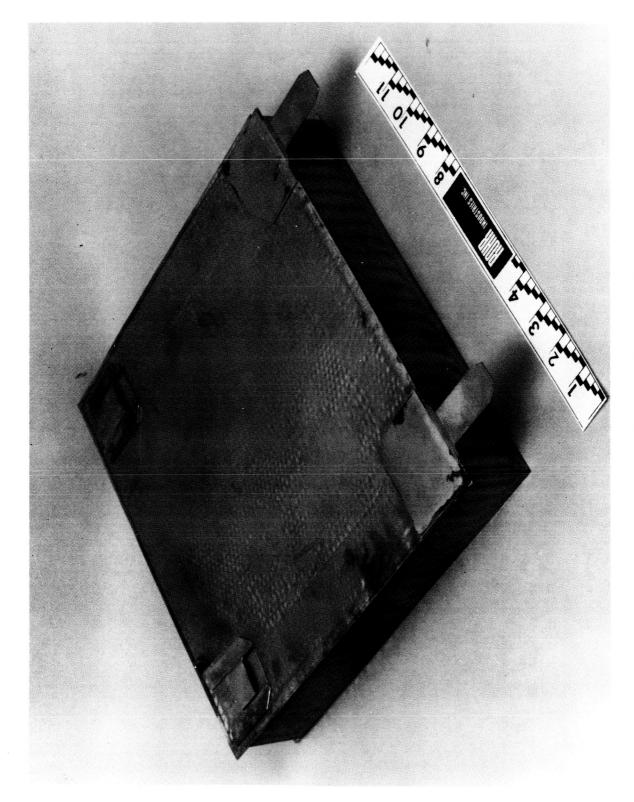


Figure 16. Bottom of Completed Bi-Metal Panel

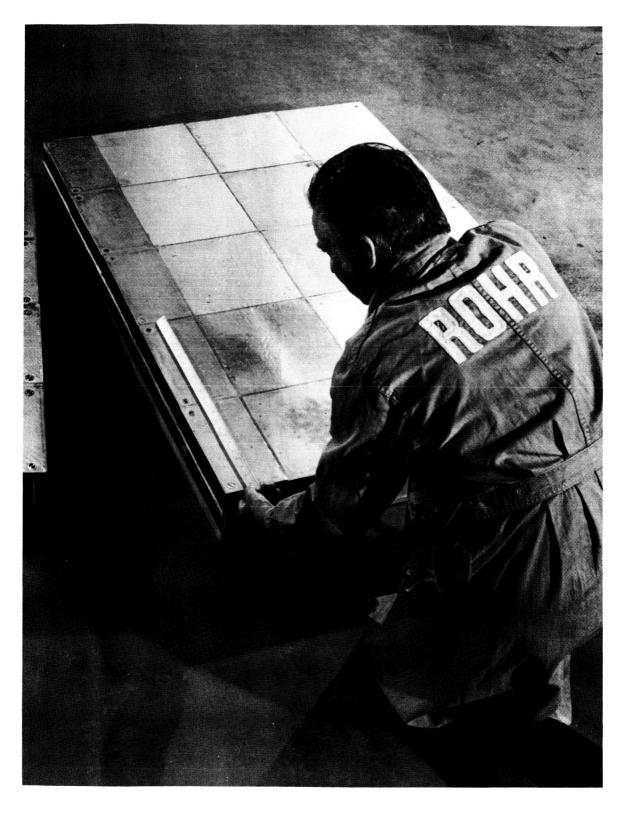
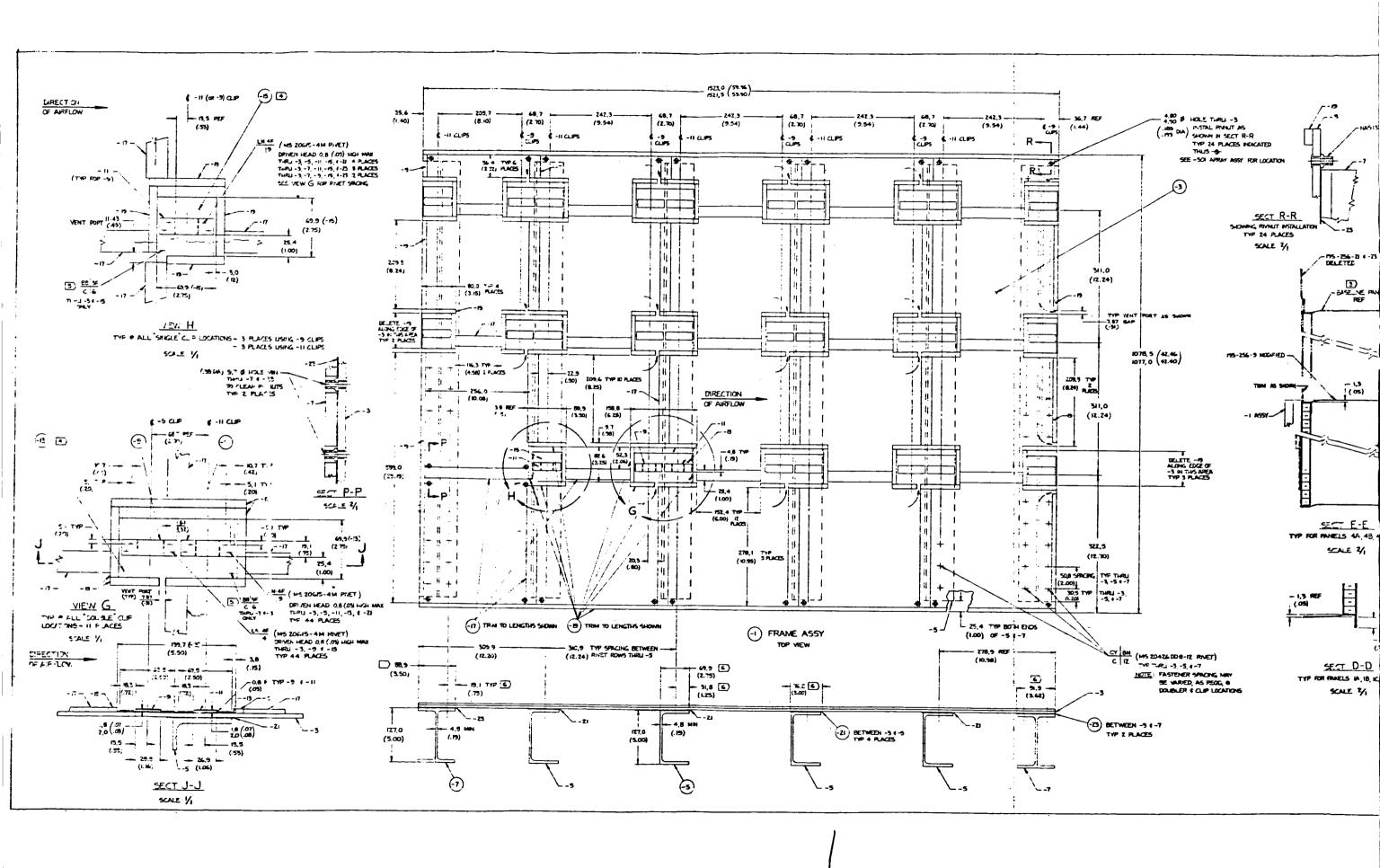
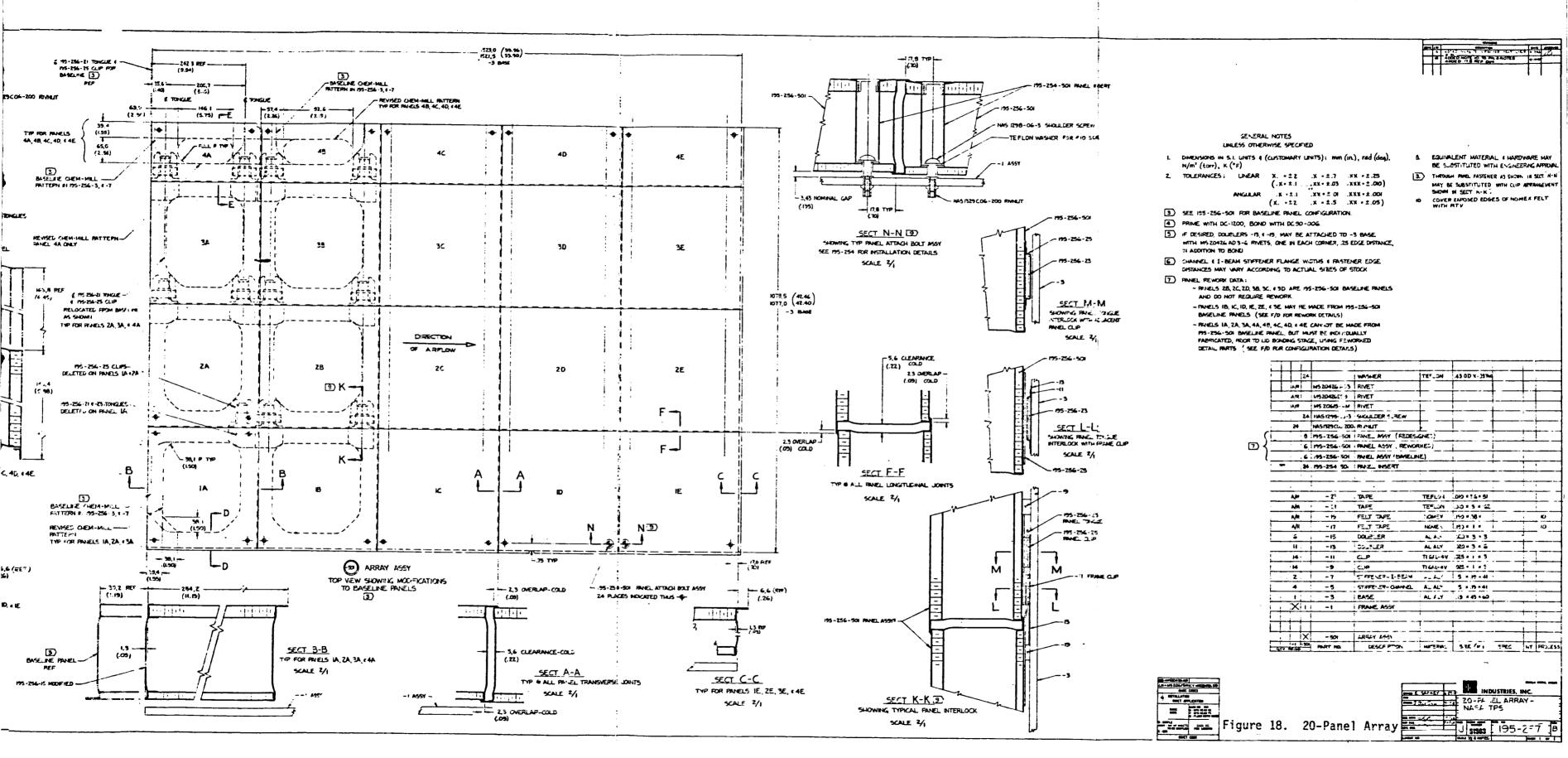


Figure 17. Superalloy--Titanium--Silica Sandwich Panel--20-Panel Array





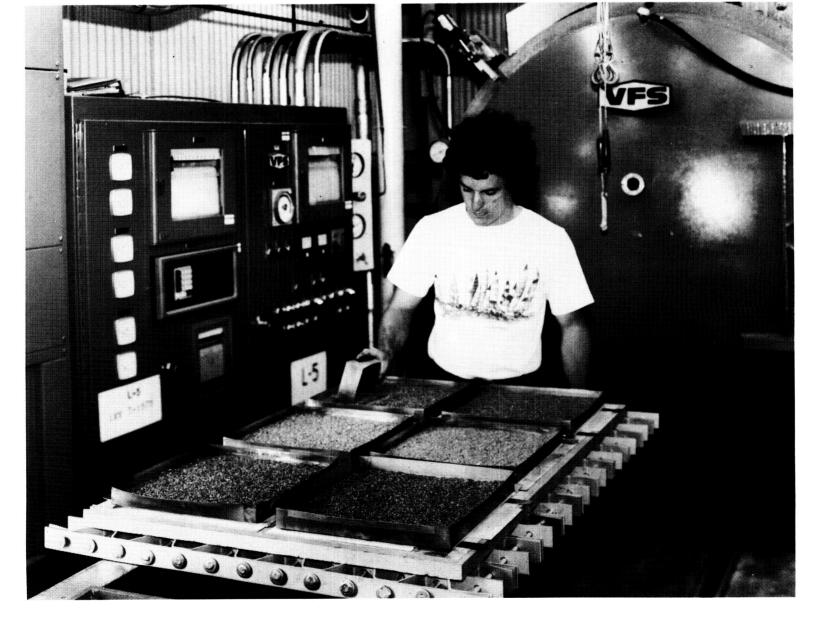


Figure 19. Six Titanium Subassemblies being Removed from Vacuum Furnace after LID Bonding

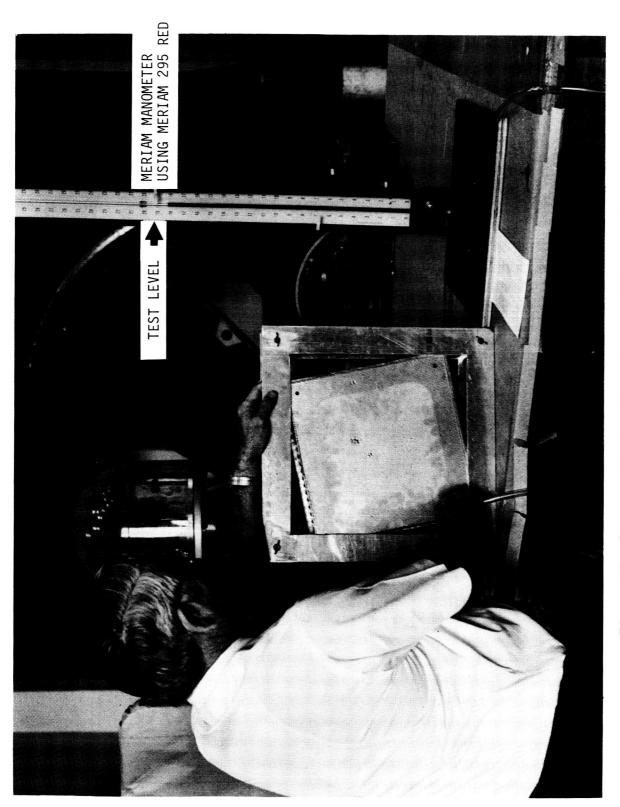


Figure 20. Pressure Testing Superalloy Sandwich Panel

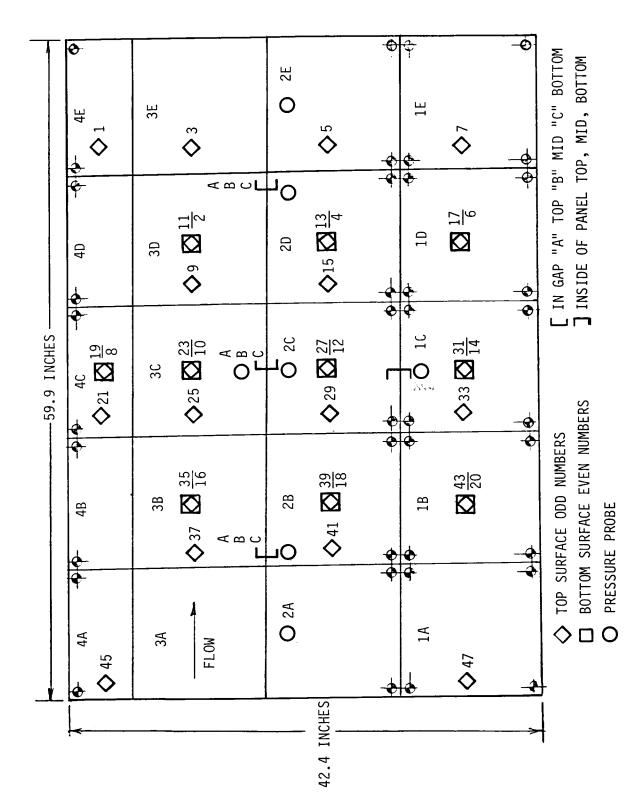


Figure 21. Thermocouple Layout by Panel Serial Number

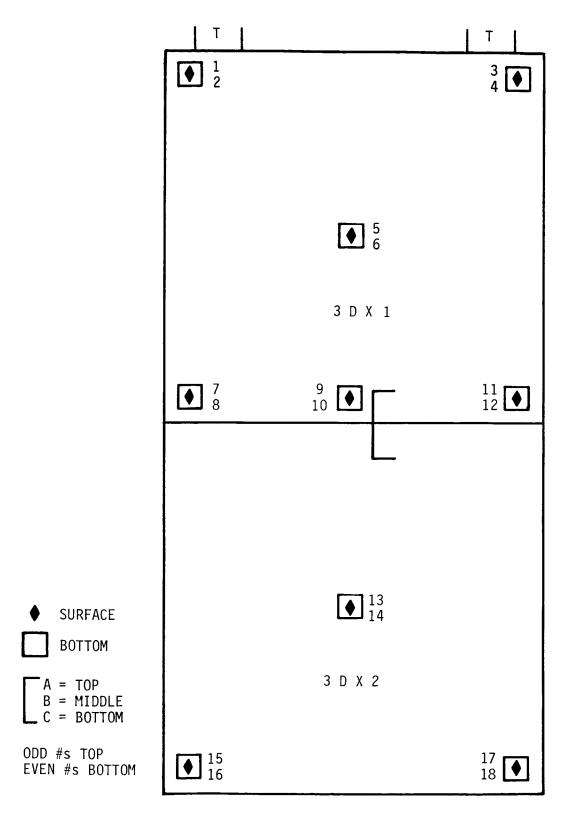


Figure 22. Thermocouple Layout for 2-Panel Array

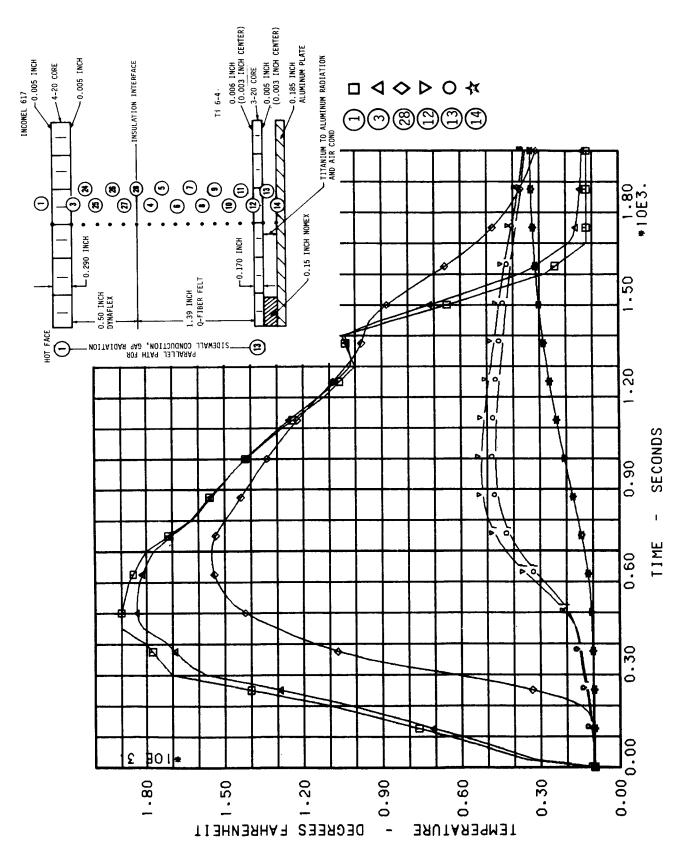


Figure 23. Thermal Math Model, Transient Analysis

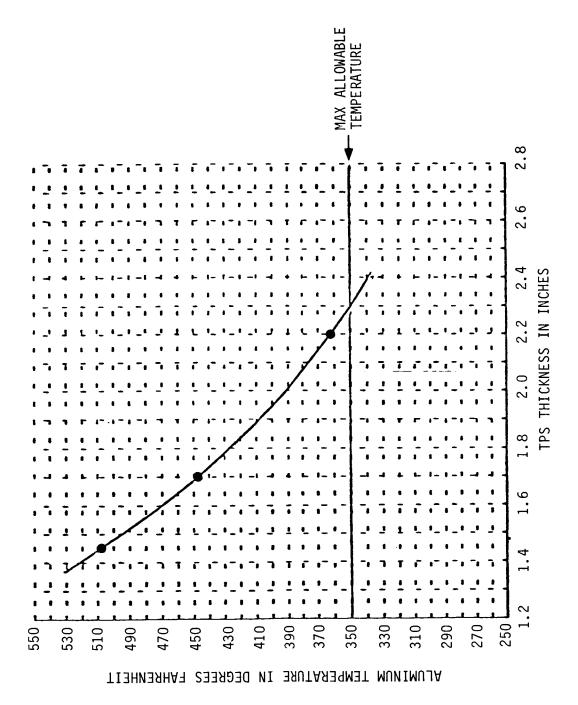


Figure 24. B.P. 1300 Aluminum Temperature Versus TPS Thickness NOMEX Thickness is Not Included



Figure 25. Guarded Hot Plate with Zoned Heating



Figure 26. Guarded Hot Plate with Thermac Controller

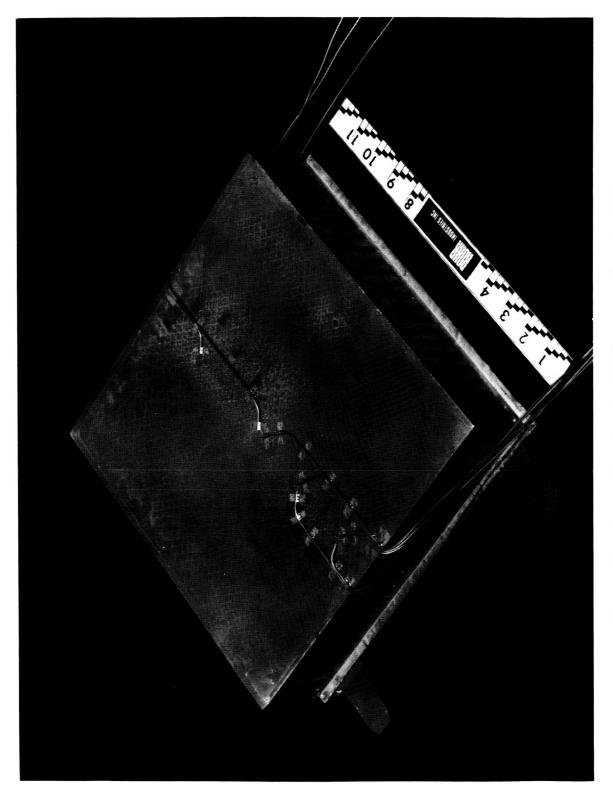


Figure 27. Superalloy Panel with Thermocouples Installed.

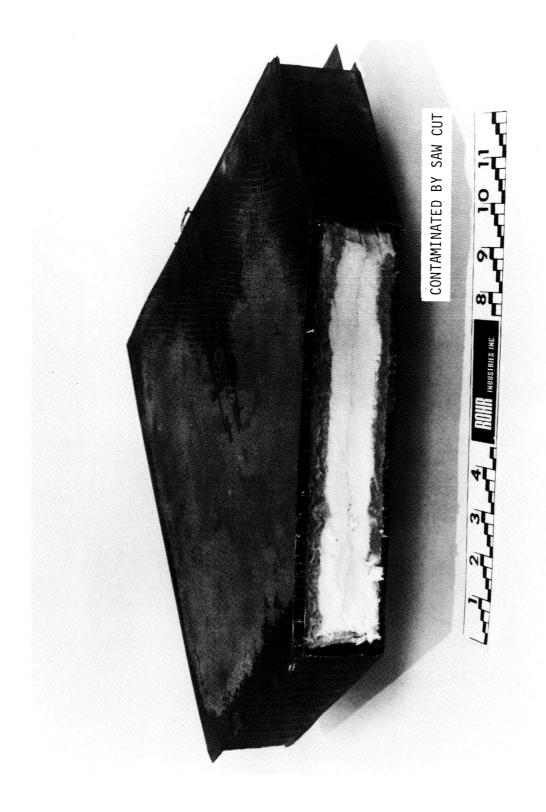


Figure 28. Superalloy Panel Section after Thermal Conductivity Tests

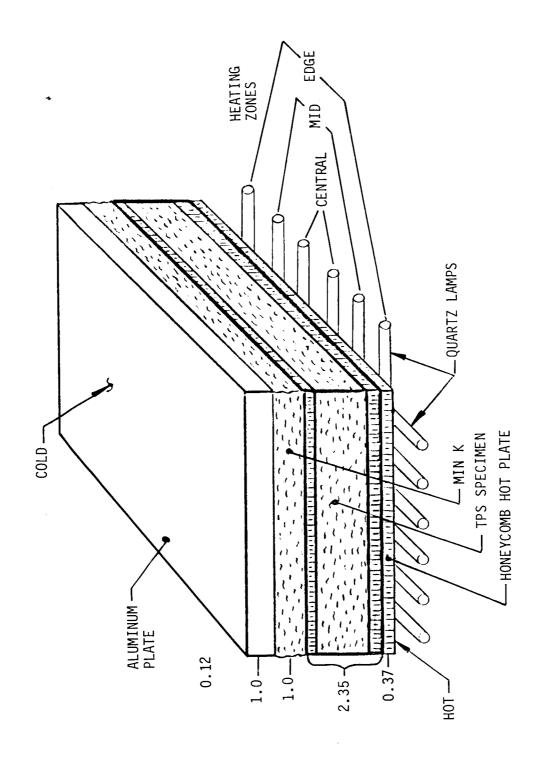


Figure 29. Thermal Conductivity Test Setup

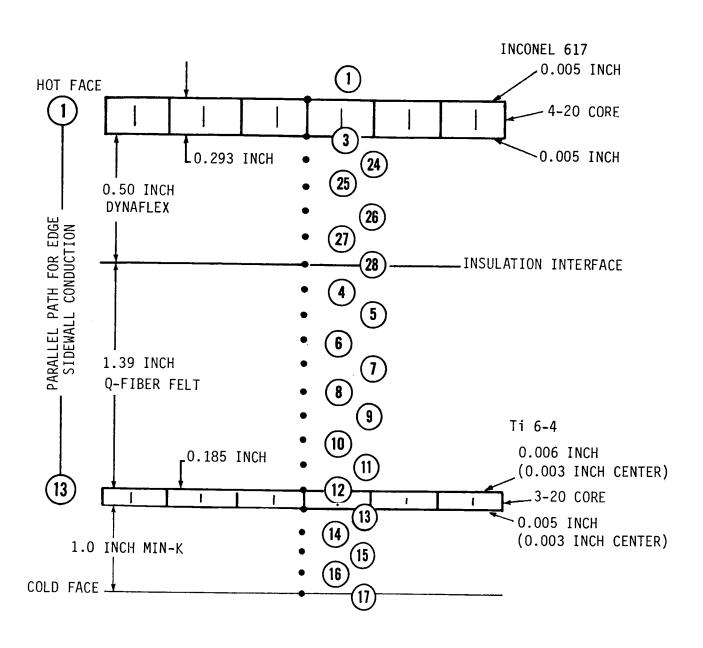


Figure 30. Thermal Math Model, Steady State Analysis

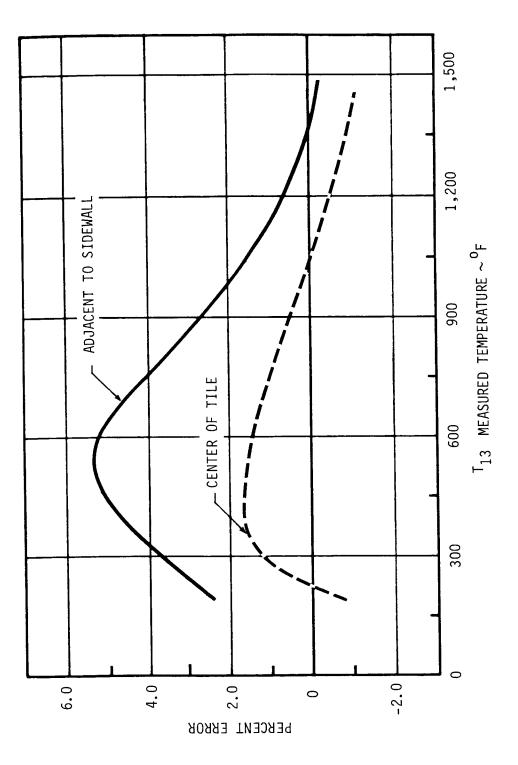


Figure 31. Percent Error Versus Measured Temperature

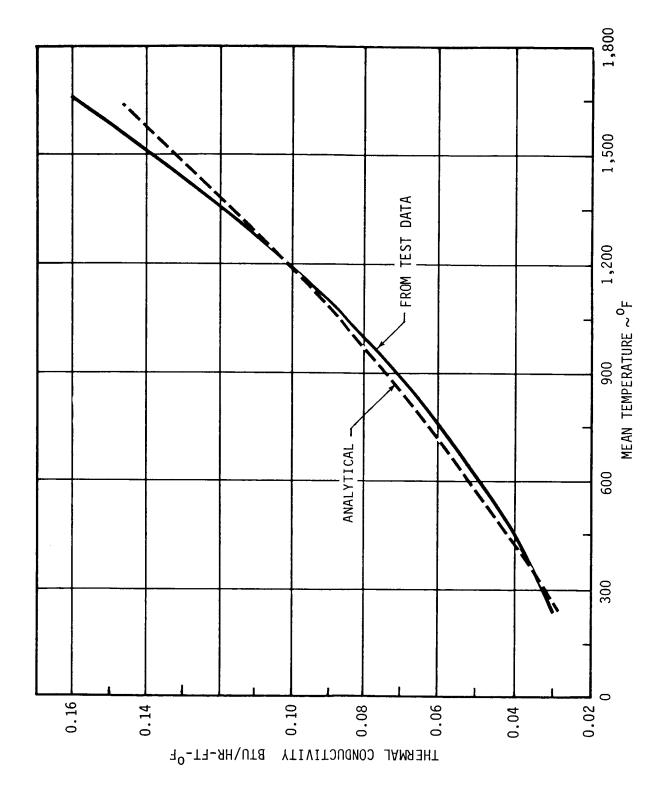
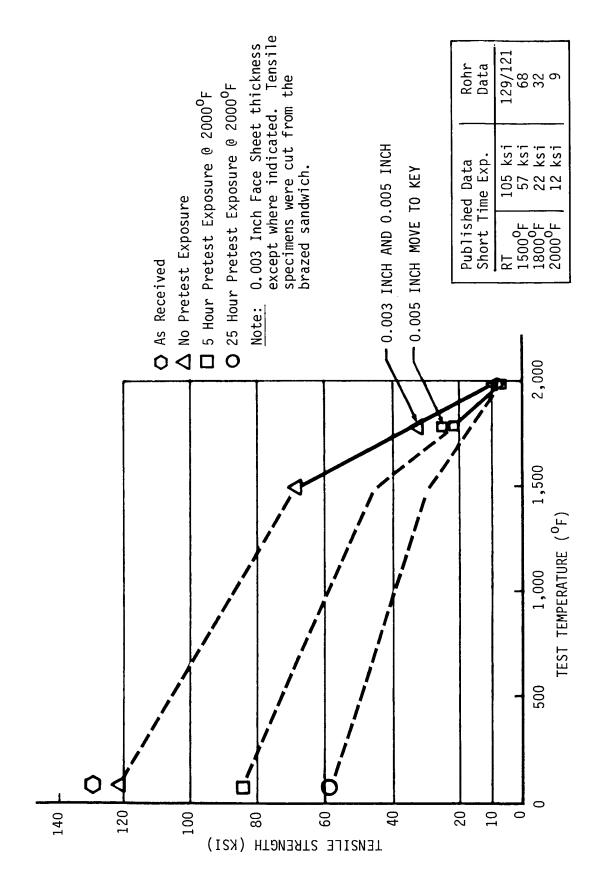
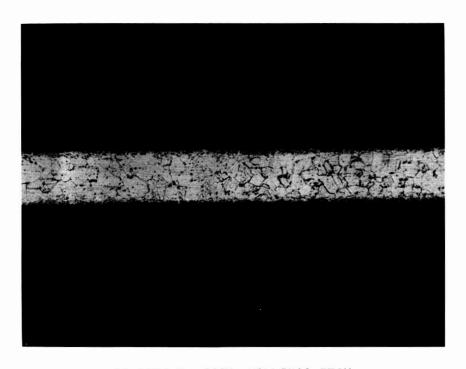


Figure 32. Effective Thermal Conductivity as a Function of Temperature



Face Sheet Ultimate Tensile Strength Versus Test Temperature INCONEL 617 Figure 33.



81-1036-5 200X KALLINGS ETCH

Figure 34. Photomicrograph of As-Received Foil



81-1036-1 200X KALLINGS ETCH

Figure 35. Photomicrograph of Brazed Foil after 5 Hours of Exposure to $2000^{\rm O}{\rm F}$



81-1036-4 200X KALLINGS ETCH

Figure 36. Photomicrograph of Brazed Foil after 25 Hours of Exposure to $2000\,^{\circ}\text{F}$

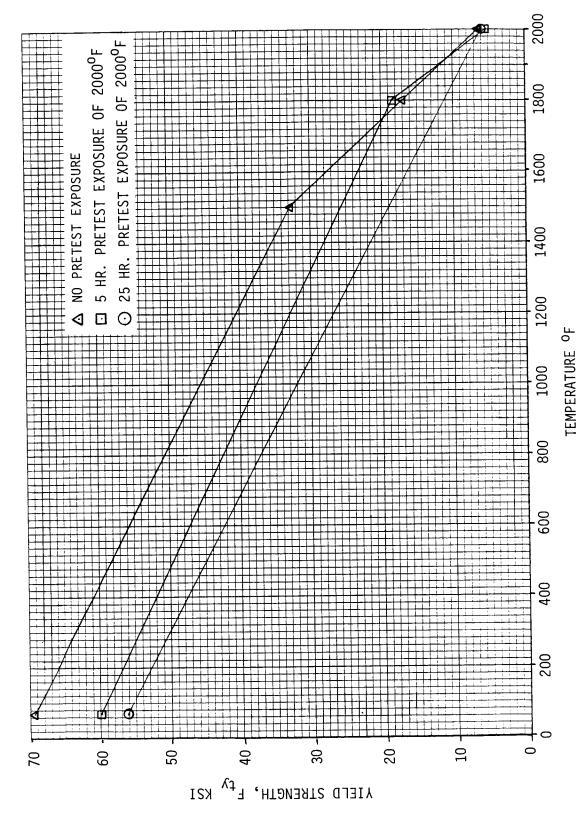
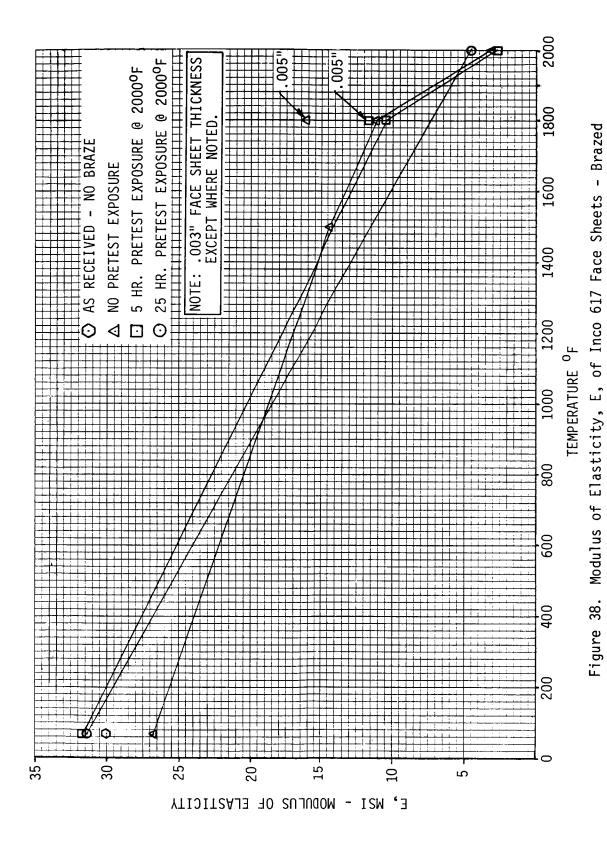


Figure 37. Tensile Yield Strength of Inco 617 Face Sheets - Brazed



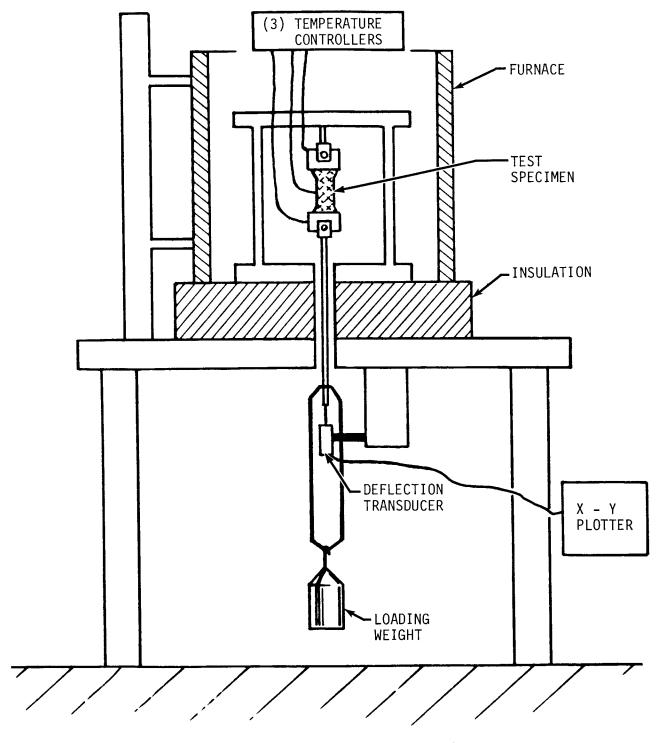


Figure 39. Schematic of Creep Test Setup

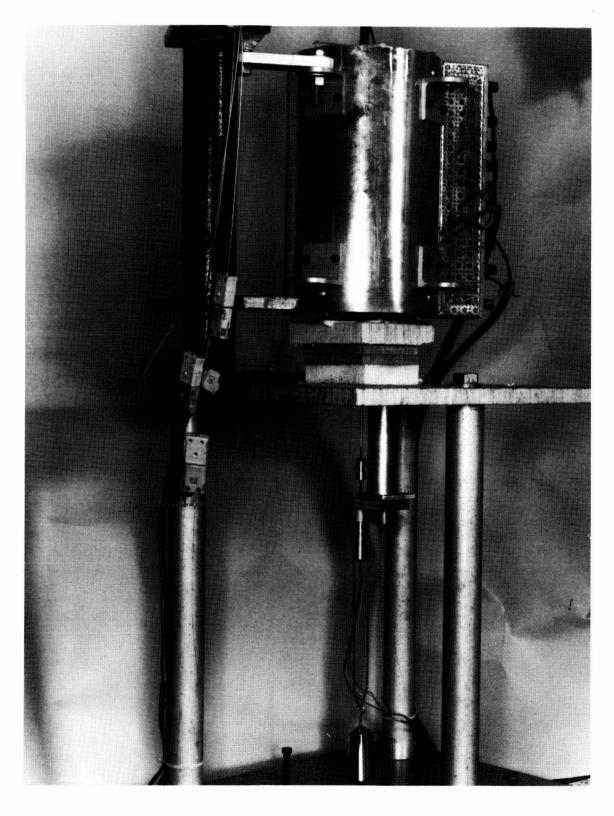


Figure 40. Overall View of Creep Test Setup

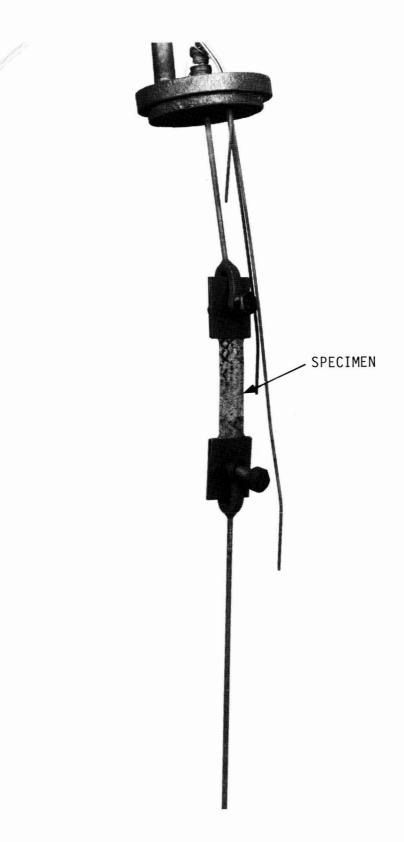


Figure 41. Large Creep Test Specimen

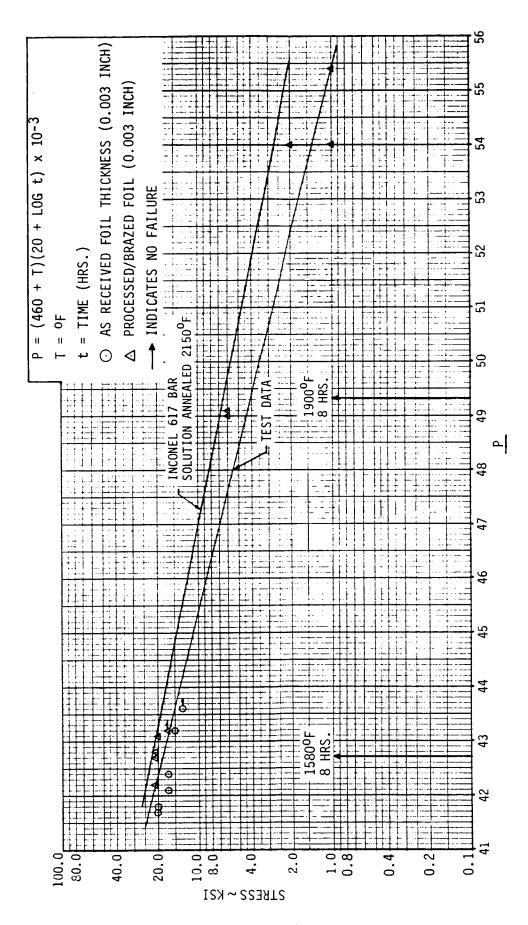


Figure 42. INCONEL 617 Creep Rupture Larson--Miller Plot

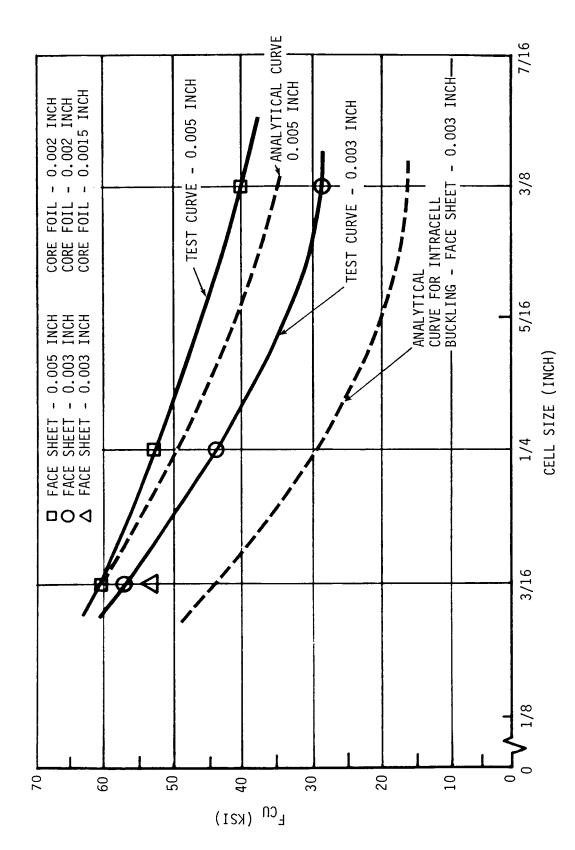


Figure 43. Edgewise Compression Strength Versus Cell Size INCONEL 617 (Room-Temp) 7.4.1 (see table 17)

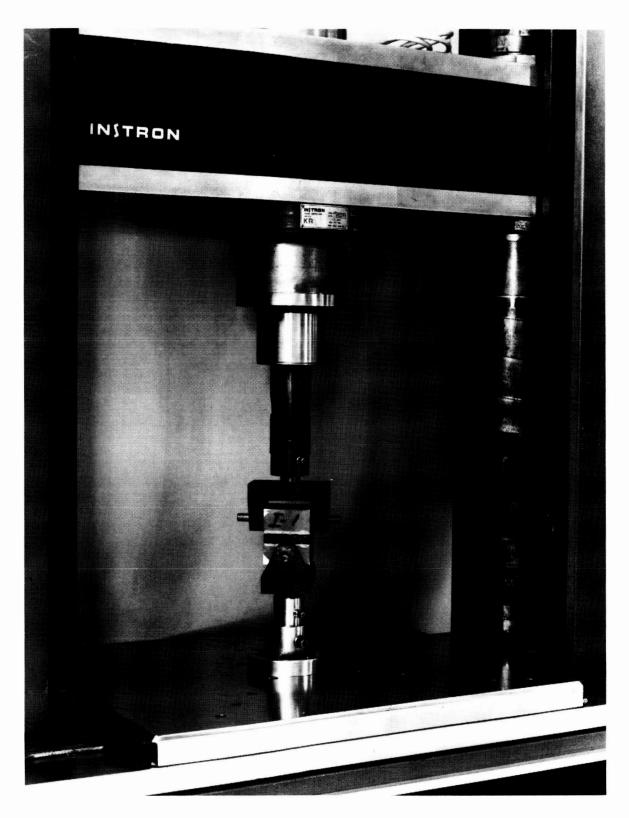


Figure 44. Room Temperature Flatwise Tension Test Setup

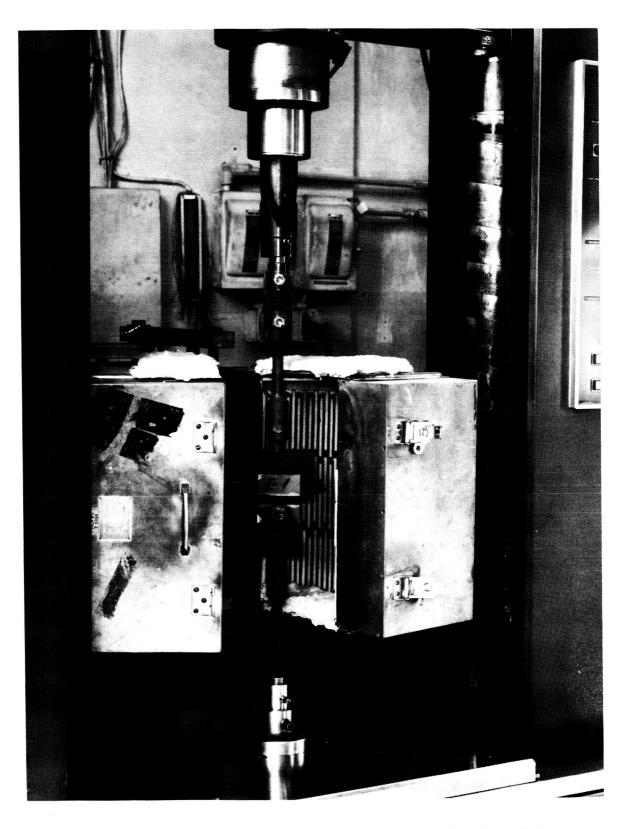


Figure 45. Elevated Temperature Flatwise Tension Test Setup

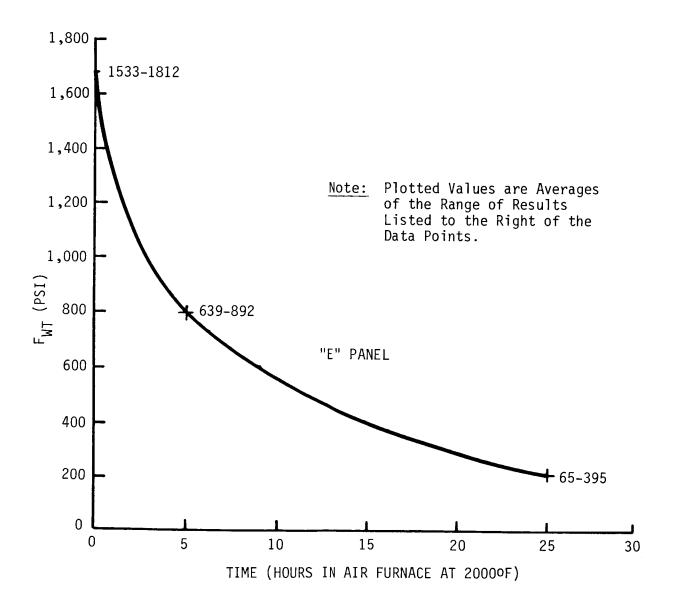
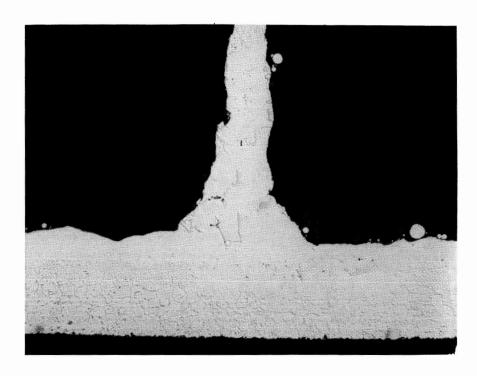
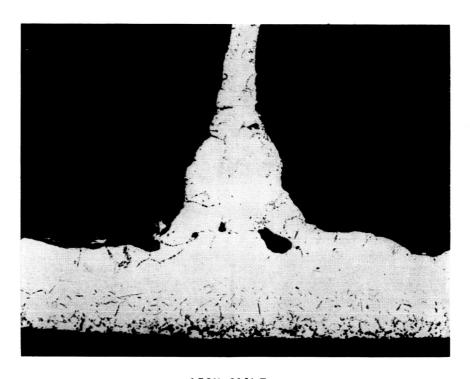


Figure 46. Flatwise Tension Strength Versus Exposure Time INCONEL 617 (Room Temperature Test)



150X SCALE

Figure 47. Photomicrograph of Brazed Sandwich Joint--No Thermal Exposure



150X SCALE

Figure 48. Photomicrograph of Brazed Sandwich Joint--5 Hours of $2000^{\circ}\mathrm{F}$ Exposure



150X SCALE

Figure 49. Photomicrograph of Brazed Sandwich Joint--25 Hours of $2000^{\circ}\mathrm{F}$ Exposure

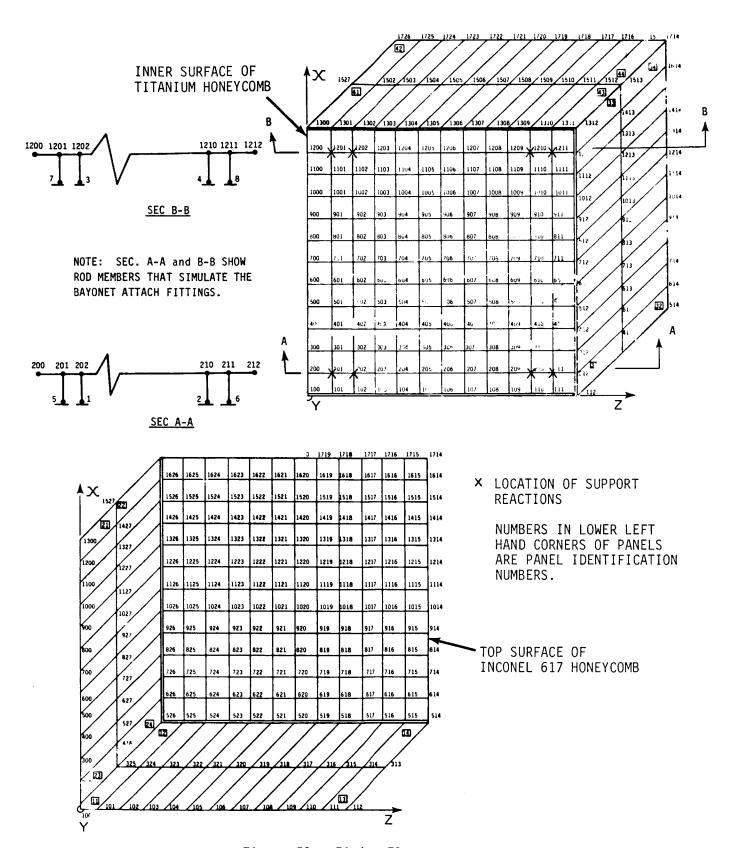
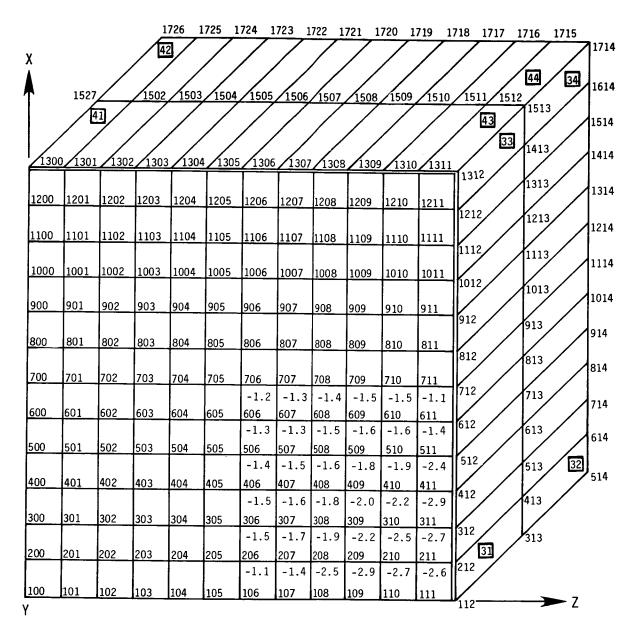


Figure 50. Finite Element Model

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1200	1226	1225	1224	1223	1222	1221	1220	1219	1218	1217	1216	1215	1214
1100	1126	1125	1124	1123	1122	1121	1120	1119	1118	1117	1116	1115	1114
	-2.5	-6.0	-7.8	-8.9	-9.5	-9.7			<u> </u>				1
1000	1026	1025	1024	1023	1022	1021	1020	1019	1018	1017	1016	1015	1014
	-3.0	-6.8	-8.5	-9.4	-9.7	-9.5							
900 927	926	925	924	923	922	921	920	919	918	917	916	915	914
	-3.5	-7.7	-9.0	-9.4	-9.4	-8.9	į						
800	826	825	824	823	822	821	820	819	818	817	816	815	814
700	-4.4	-8.2	-9.0	-9.0	-8.5	-7.8		L		ļ			
727	726 -5.9	725	724	723	722	721	720	719	718	717	716	715	714
600	626	-8.3	-8.2	-7.7	-6.8	-6.0	600	610	610	c			
627	-5.2	625 -5.9	624	623	622	621	620	619	618	617	616	615	614
500	526	525	-4.4 524	-3.5 523	-3.0 522	-2.5 521	520	519	518	517	516	515	514
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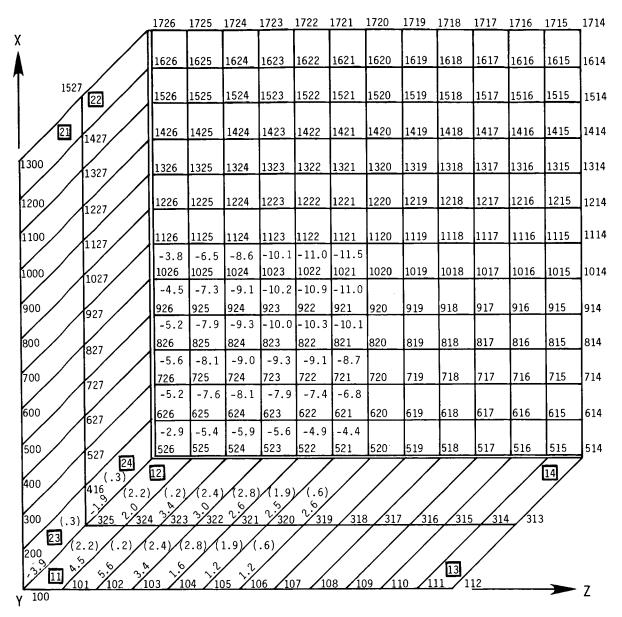
- 1. NUMBERS IN LOWER LEFT HAND CORNERS OF PANELS ARE PANEL IDENTIFICATION NUMBERS.
- 2. NUMBERS IN MIDDLE OF PANELS ARE PRINCIPAL STRESS VALUES IN UNITS OF KSI.
- 3. (+) ARE TENSION. (-) ARE COMPRESSION.
- 4. NUMBERS IN () ARE SHEAR STRESSES.

Figure 51-A. Ascent Crush Pressure Condition Top Surface of Panel \sim Inconel Honeycomb - Stress Output



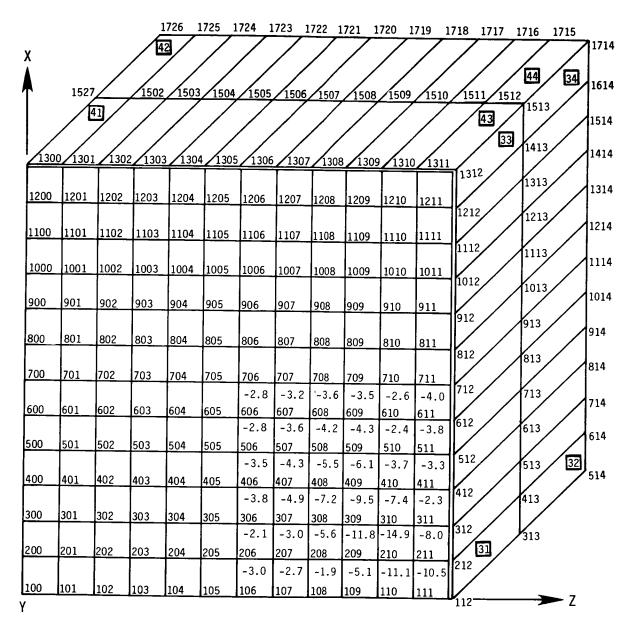
- 1. NUMBERS IN LOWER LEFT HAND CORNERS OF PANELS ARE PANEL IDENTIFICATION NUMBERS.
- 2. NUMBERS IN MIDDLE OF PANELS ARE PRINCIPAL STRESS VALUES IN UNITS OF KSI.
- 3. (+) ARE TENSION.
 - (-) ARE COMPRESSION.
- 4. NUMBERS IN () ARE SHEAR STRESSES.

Figure 51-B. Ascent Crush Pressure Condition Bottom Surface of Panel ~ Titanium Honeycomb - Stress Output



- 1. NUMBERS IN LOWER LEFT HAND CORNERS OF PANELS ARE PANEL IDENTIFICATION NUMBERS.
- 2. NUMBERS IN MIDDLE OF PANELS ARE PRINCIPAL STRESS VALUES IN UNITS OF KSI.
- 3. (+) ARE TENSION.
 - (-) ARE COMPRESSION.
- 4. NUMBERS IN () ARE SHEAR STRESSES.

Figure 51-C. Ascent Blowoff Pressure Condition Top Surface of Panel ~ Inconel Honeycomb - Stress Output



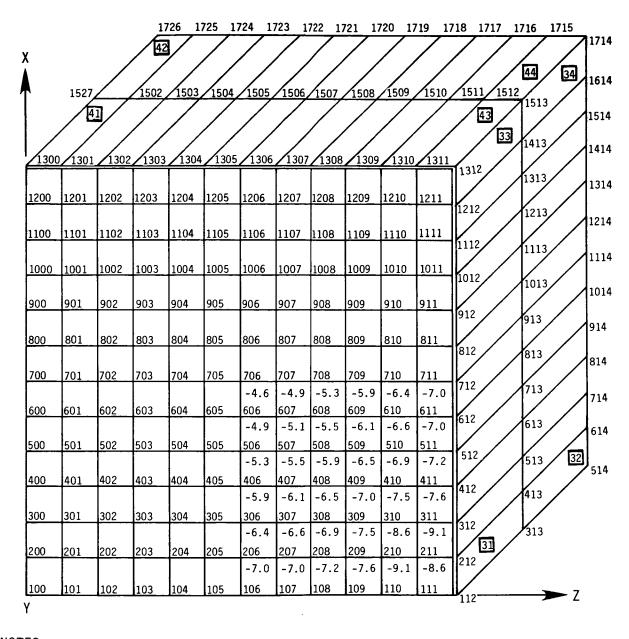
- 1. NUMBERS IN LOWER LEFT HAND CORNERS OF PANELS ARE PANEL IDENTIFICATION NUMBERS.
- 2. NUMBERS IN MIDDLE OF PANELS ARE PRINCIPAL STRESS VALUES IN UNITS OF KSI.
- 3. (+) ARE TENSION.
 - (-) ARE COMPRESSION.
- 4. NUMBERS IN () ARE SHEAR STRESSES.

Figure 51-D. Ascent Blowoff Pressure Condition Bottom Surface of Panel ~ Titanium Honeycomb - Stress Output

	1726	1725	1724	1723	1722	1721	1720	1719	1718	1717	1716	1715	1714
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	1626	1625	1624	1623	1622	1621	1620	1619	1618	1617	1616	1615	1614
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	1426	1425	1424	1423	1422	1421	1420	1419	1418	1417	1416	1415	1414
1427	1	1120											
1300	1326	1325	1324	1323	1322	1321	1320	1319	1318	1317	1316	1315	1314
1200	1226	1225	1224_	1223_	1222	1221	1220_	1219	1218	1217	1216	1215	1214
1100	1126	1125	1124	1123	1122	1121	1120	1119_	1118	1117	1116	1115	1114
	-6.6 1026	-5.8 1025	-4.9 1024	-4.1 1023	-3.7 1022	-3.4 1021	1020	1019	1018	1017	1016	1015	1014
1000	-6.8	-6.0	-5.1	-4.4	-3.9	-3.7	1						1
900	926	925	924	923	922	921	920_	919	918	917	916	915	914
	7.1	-6.6	-5.6	-4.9	-4.4	-4.1							
800 827	826	825	824	823	822	821	820_	819	818	817	816	815	814
700	726	-7.4 725	-6.4 724	-5.6 723	-5.1 722	- 4.9 721	720	719	718	717	716	715	714
727	-9.1	-8.7	-7.4	-6.6	-6.0	-5.8]
600	626	625	624	623	622	621	620_	619	618	617	616	615	614
	-8.7 526	-9.1 525	-7.7 524	-7.1 523	-6.8 522	-6.6 521	520	519	518	517	516	515	514
	12/											14	7
416 (4.0	_/	(1.8)	(1.1)	(.6)	(.2)	/ ,	/ ,	/ ,	/ ,	/ ,	/ /		
300 325 324 323 322 321 320 319 318 317 316 315 314 313													
23 (4.0) (2.9) (1.8) (1.1) (.6) (.4)													
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- 1. NUMBERS IN LOWER LEFT HAND CORNERS OF PANELS ARE PANEL IDENTIFICATION NUMBERS.
- 2. NUMBERS IN MIDDLE OF PANELS ARE PRINCIPAL STRESS VALUES IN UNITS OF KSI.
- 3. (+) ARE TENSION. (-) ARE COMPRESSION.
- 4. NUMBERS IN () ARE SHEAR STRESSES.

Figure 51-E. Descent Condition Top Surface of Panel \sim Inconel Honeycomb - Stress Output



- 1. NUMBERS IN LOWER LEFT HAND CORNERS OF PANELS ARE PANEL IDENTIFICATION NUMBERS.
- 2. NUMBERS IN MIDDLE OF PANELS ARE PRINCIPAL STRESS VALUES IN UNITS OF KSI.
- 3. (+) ARE TENSION.
 - (-) ARE COMPRESSION.
- 4. NUMBERS IN () ARE SHEAR STRESSES.

Figure 51-F. Descent Condition Bottom Surface of Panel ~ Titanium Honeycomb - Stress Output

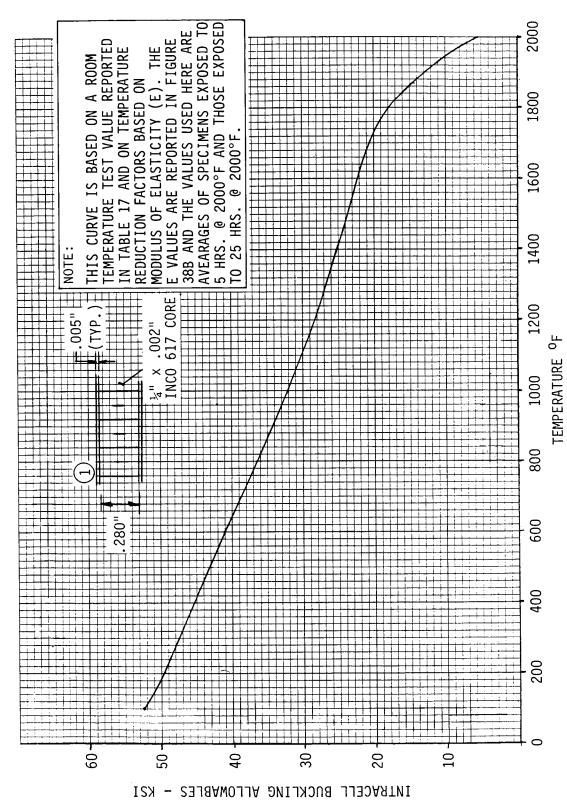
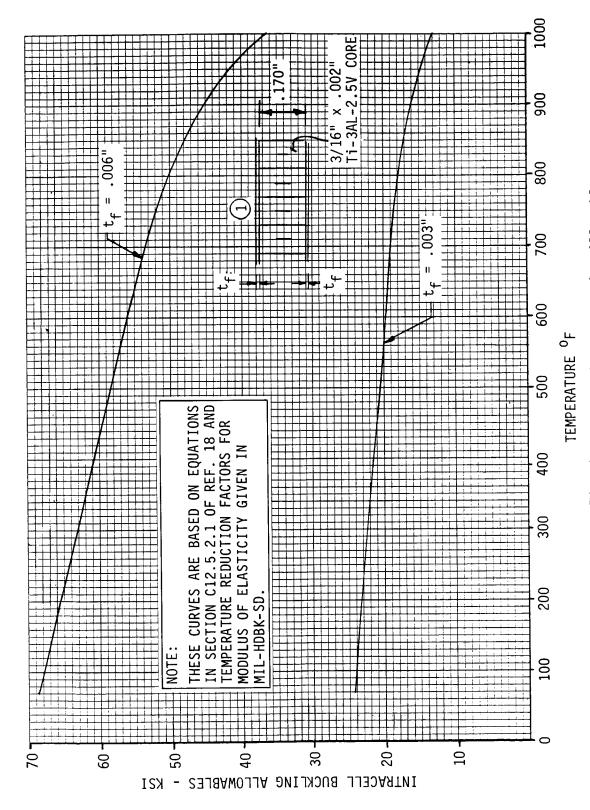


Figure 52. Inconel Honeycomb Compression Allowables



igure 53. Titanium Honeycomb Compression Allowables

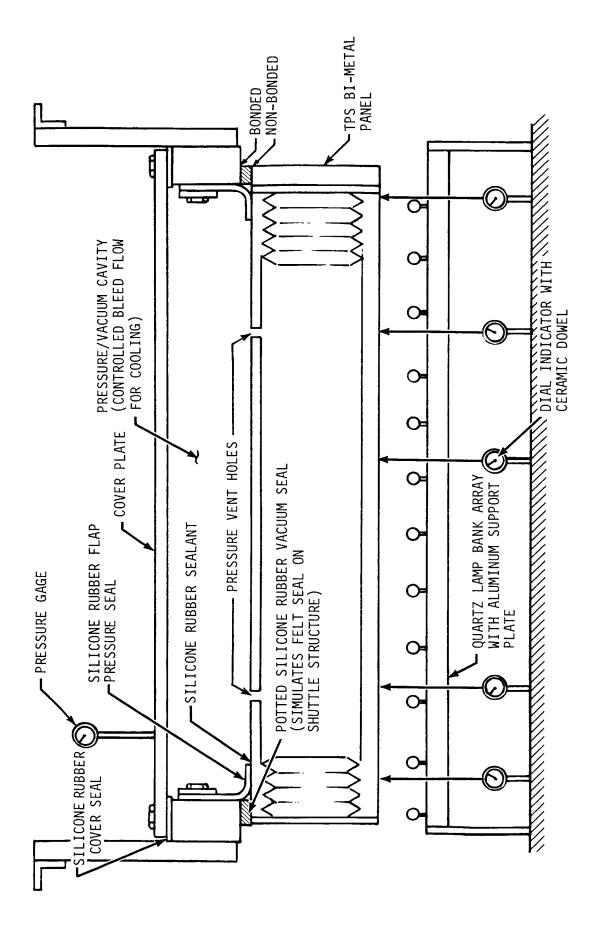


Figure 54. Schematic of Text Fixture for Thermal/Pressure Gradient Tests

Figure 55. Test Apparatus for Pressure Testing with Thermal Gradient



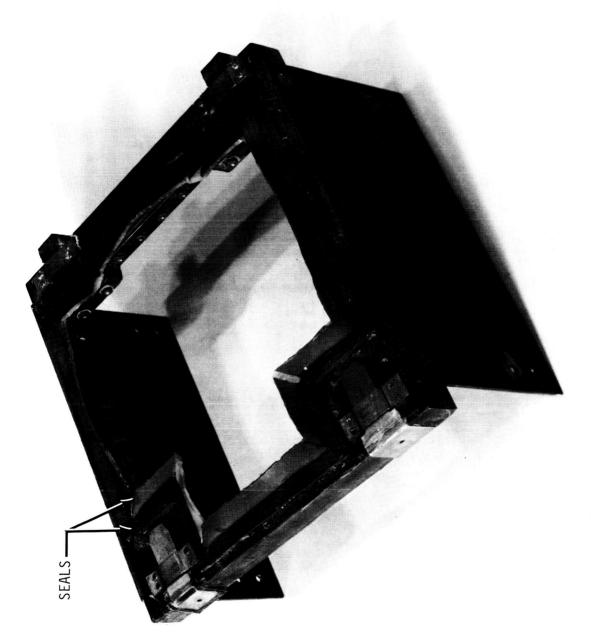


Figure 57. Test Chamber in Inverted Position

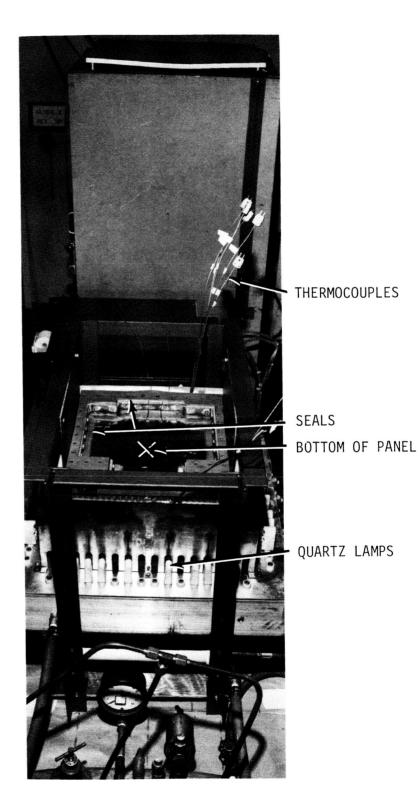


Figure 58. Pressure Test Apparatus

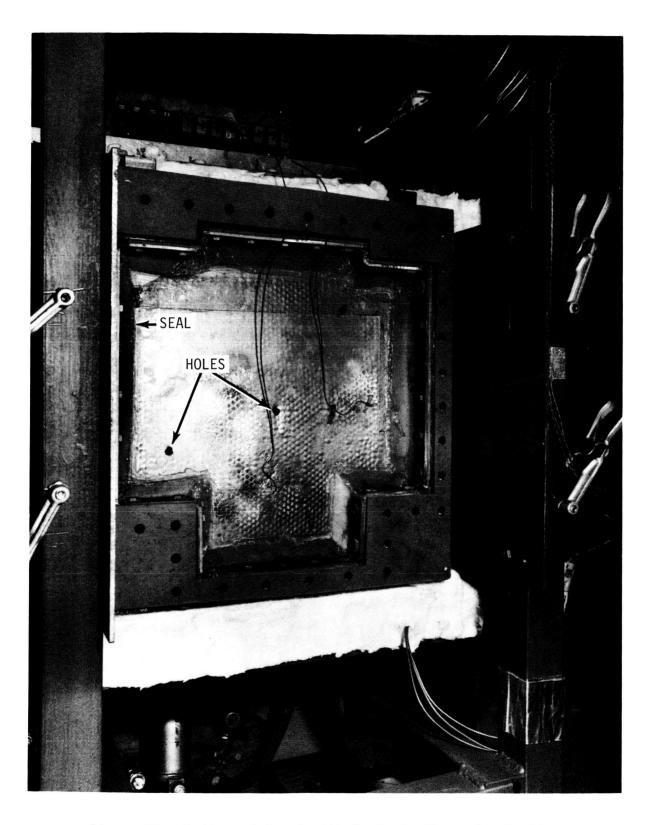
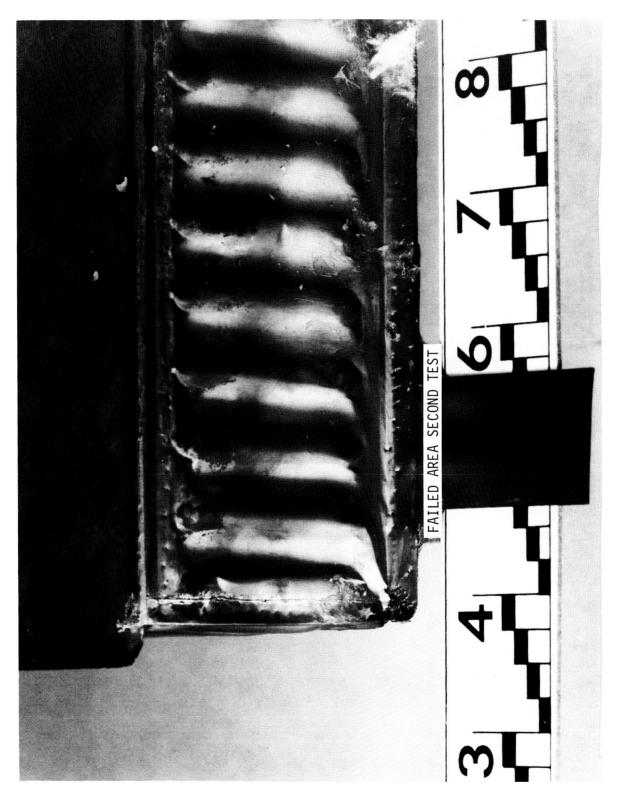


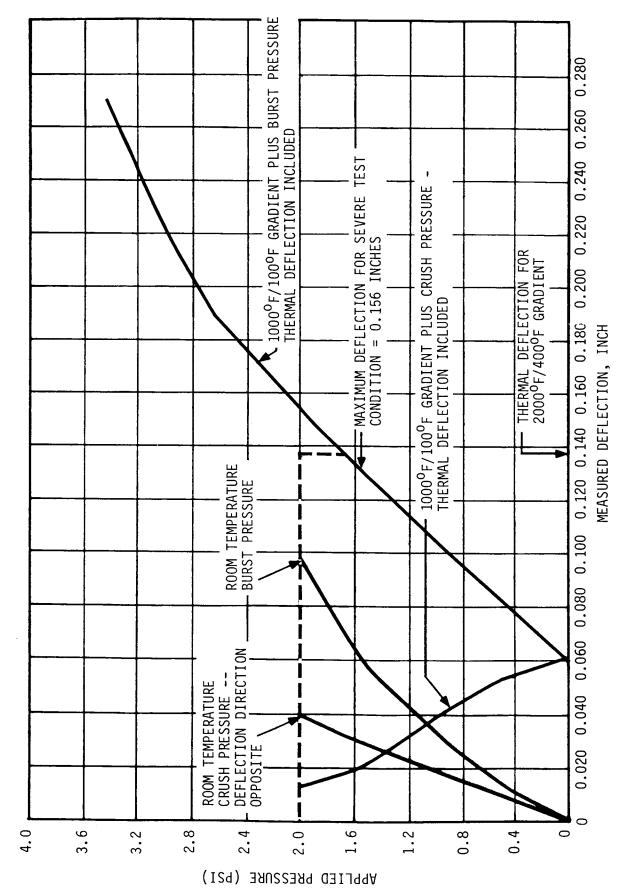
Figure 59. Bottom of Panel with Seals in Place for Testing



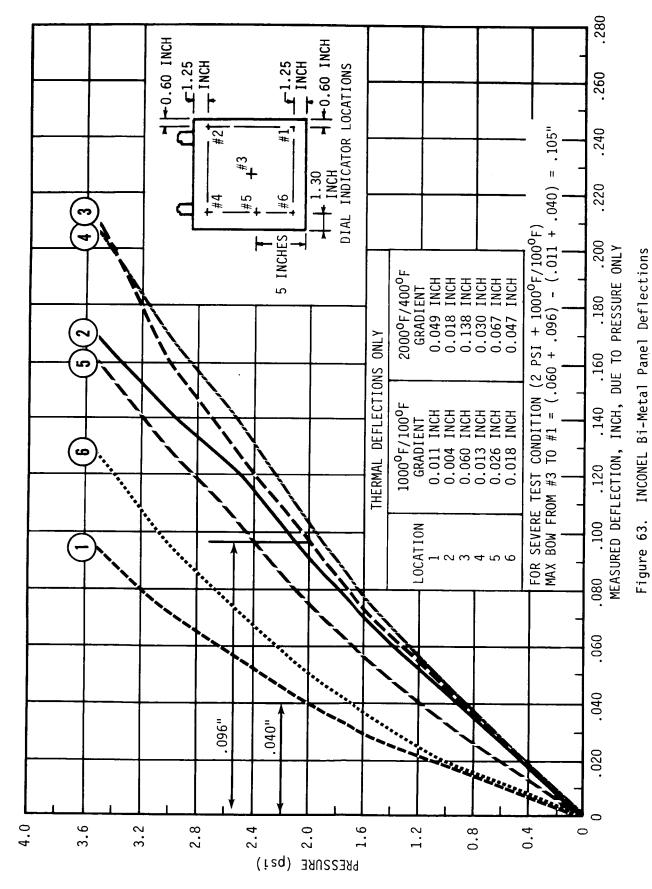
Panel Failed at 3.6 psi in the INCONEL 617 Material at the Bi-Metallic Joints During the Second Test, View 1 Figure 60.



Panel Failed at 3.6 psi in the INCONEL 617 Material at the Bi-Metallic Joint During the Second Test, View $2\,$ Figure 61.



INCONEL 617, Ti-6-4, Silica Fiber Sandwich Panel Applied Pressure (psi) Versus Center Panel Deflections (inch) for Various Loading Conditions Figure 62.



APPENDIX

TPS THREE DIMENSIONAL "NASTRAN"
FINITE ELEMENT MODEL

THE FOLLOWING PAGES CONTAIN CODED SAMPLE
OF TPS PANEL FOR THREE DIMENSIONAL
"NASTRAN" FINITE ELEMENT ANALYSIS.

SAMPLE INPUT REPRESENTS BOTH PRESSURE
AND THERMAL STATIC ANALYSIS.

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